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# Direct Imaging of Atomic-scale Surface Structures of Brookite TiO<sub>2</sub> Nanoparticles by Frequency Modulation Atomic Force Microscopy in Liquid

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#### Abstract

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Direct imaging of atomic-scale surface structures of  $TiO_2$  nanoparticles is a promising method for understanding the detailed mechanism of their photocatalytic activities. Atomic force microscopy (AFM) is one of the analytical tools used for direct imaging of surfaces, but atomic-scale AFM studies of  $TiO_2$  nanoparticles in a liquid have not been reported. Here, we report two methods for fixing of  $TiO_2$  nanocrystals: an electrostatic method and a cross-linking method. Both methods enabled visualization of subnanoscale surface structures of brookite  $TiO_2$  nanoparticles in a liquid by frequency modulation AFM (FM-AFM). The FM-AFM imaging results and density functional theory molecular dynamics calculations suggest that the subnanoscale structures in the FM-AFM images can be explained by the arrangement of the surface topmost atoms or their hydration structures on the brookite  $TiO_2$  (210) surface. Our results open up various possibilities for studying atomic-scale surface structures of nanoparticles in liquids by AFM.

## Introduction

Titanium dioxide  $(TiO_2)$  is attracting attention as a photocatalyst for many applications 16 such as environmental cleanup<sup>1,2</sup> and the synthesis of useful chemicals.<sup>3,4</sup> The thermal sta-17 bility and the photocatalytic activity of  $TiO_2$  in water broadens its potential range of prac-18 tical applications. Electron microscopy and X-ray diffraction are the main techniques used 19 for investigating the crystal structures of TiO<sub>2</sub>.<sup>5-14</sup> Although such techniques can be used 20 to determine precise crystal structures and their orientations, they cannot provide detailed 21 structural information on surface structures. Investigations of the surfaces of  $TiO_2$  are impor-22 tant for understanding the mechanisms of photocatalytic reactions. Scanning probe micro-23 copy (SPM) techniques such as scanning tunneling microscopy and atomic force microscopy 24 (AFM) enable visualization of the atomic-scale surface structures of various crystals, as pre-25 viously reported.  $^{15,16}$  Rutile, anatase, and brook ite are the major stable polymorphs of  ${\rm TiO}_2$ 26 cystals.<sup>11,17–19</sup> Early SPM studies of TiO<sub>2</sub> were mainly performed on crystals of rutile, which is the most stable TiO<sub>2</sub> crystal phase.<sup>20–37</sup> In addition, there have been several reports of atomic-scale investigations of anatase TiO<sub>2</sub> by AFM in a vacuum.<sup>38</sup> However, the surface structure of brookite TiO<sub>2</sub> has not yet been experimentally investigated by SPM techniques. The atomic-scale surface structures expected from the bulk x-ray crystallography or MD simulation have not been confirmed by any microscopic experiments.

One of the major reasons for the lack of SPM investigations of brookite  $TiO_2$  sur-7 faces is that high-quality crystals of brookite  $TiO_2$  are more difficult to prepare than ru-8 tile and anatase crystals. Recently, methods for preparing high-purity brookite nanocrysq tals have been established by using appropriate precursors, additives and synthetic condi-10 tions.<sup>6,8–10,12–14,39,40</sup> Methods for controlling the particle shape and exposed facets of brookite 11  $TiO_2$  nanocrystals have also been developed for improving the photocatalytic activity.<sup>5</sup> An 12 understanding of the atomic-scale surface structures of brookite  $TiO_2$  is needed to clarify the 13 relationship between the surface structures and the photocatalytic activity. The development 14 of methods for synthesizing high-quality crystals of brookite  $TiO_2$  has enabled examination 15 of surfaces by SPM techniques. Investigation of the surface structures of brookite  $TiO_2$  at 16 an atomic resolution should contribute to our understanding of the mechanisms of photo-17 catalytic reactions. 18

In this study, a laboratory-built apparatus for frequency modulation atomic force mi-19 croscopy (FM-AFM)<sup>41-43</sup> was used to investigate the atomic-scale surface structures of 20 brookite  $TiO_2$  in water. FM-AFM imaging in liquids has been used for various samples, 21 ranging from inorganic crystals  $^{44,45}$  to biomolecules.  $^{46,47}$  As mentioned above, atomic-scale 22 AFM investigations of rutile and anatase  $TiO_2$  in a vacuum have already been reported. 23 However, although AFM images with atomic-scale striped contrasts obtained at the rutile 24 (110) surface in a liquid have been reported, <sup>48</sup> the contrasts within each stripe were not clear 25 enough to allow their detailed comparison with an atomic surface model. To the best of our 26 knowledge, there has been no report of atomic-scale AFM studies of  $TiO_2$  surfaces in liquids, 27

not only of brookite, but also of rutile and anatase.

The spatial resolution of previously reported FM-AFM images obtained in liquids sug-2 gests that atomic-scale visualization of brookite TiO<sub>2</sub> surfaces is a promising approach. 3 However, there are practical problems to be overcome. At present, brookite  $TiO_2$  can be 4 prepared only in the form of nanoparticles, but atomic-scale SPM imaging of nanometer-5 sized particles is generally challenging. This is because nanocrystals are often removed from 6 the substrate by contact with the tip during scanning. In particular, the use of a stiff can-7 tilever and a small-amplitude cantilever vibration, which are required for atomic-resolution 8 FM-AFM imaging, makes it difficult to avoid this problem. To suppress the desorption of q brookite  $TiO_2$  nanoparticles, they need to be fixed on a substrate by strong interactions. 10 AFM samples that consist of micrometer-sized particles can be fixed with glue.<sup>49</sup> However, 11 this is not applicable to nanoparticles because they become buried in the glue layer. So far, 12 successful atomic-scale imaging of nanoparticles has been reported only for ionic crystals 13 grown on a substrate.<sup>50</sup> However, this is not applicable to  $TiO_2$  nanoparticles because they 14 are insoluble in water. 15

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In this study, we investigated the atomic-scale surface structure of brookite TiO<sub>2</sub> nanocrys-16 tals by FM-AFM in a liquid. We established two fixing methods for atomic-scale FM-AFM 17 imaging of  $TiO_2$  nanocrystals: an electrostatic method and a cross-linking method. Using 18 two fixing methods, direct visualizations of atomic-scale features at brookite/water interfaces 19 on nanoparticles were achieved. The obtained atomic-scale FM-AFM images allowed us to 20 precisely compare them with the atomic-scale surface model. In addition, density functional 21 theory molecular dynamics (DFT MD) simulations of the brookite TiO<sub>2</sub>-water interface were 22 performed to clarify the origin of the atomic-scale contrasts in FM-AFM images obtained in 23 a liquid. 24

## Experimental

### Synthesis of brookite TiO<sub>2</sub> nanorods

Brookite nanorods were prepared by a previously reported procedure.  $^{51,52}$  Titanium (IV) 3 ethoxide (10 g) was dissolved in ethanol (50 mL) and then deionized water (3 mL) was 4 added to the solution. The solution was stirred for 30 min and then precipitate was collected 5 by centrifugation. Amorphous  $TiO_2$  was obtained by drying the collected precipitates in a 6 vacuum oven. Amorphous TiO<sub>2</sub> (4 g) was added to a solution containing 30% H<sub>2</sub>O<sub>2</sub> (40 mL) 7 and 28% NH<sub>3</sub> (10 mL) under cooling with water. The dispersion was stirred for 2 h, and then 8 glycolic acid (2 g) was added. The dispersion changed color yellow to red. Stirring for 24 h 9 to eliminate excess  $H_2O_2$  and  $NH_3(aq)$  gave a glycolated  $TiO_2$  complex solution. The volume 10 and pH of the complex solution were adjusted to 50 mL and  $10 \text{ with } \text{NH}_3$  and deionized water. 11 The solution was placed in a Teflon-lined autoclave reactor. The reactor was heated in an 12 oven at 200 °C for 25 h. The solid substrate was separated by centrifugation, washed several 13 times with deionized water, and dried in a vacuum oven. Crystal structures of the prepared 14 brookine TiO<sub>2</sub> nanorods were confirmed with X-ray diffraction (XRD) measurement and 15 SEM imaging (see Figures S1 and S2). 16

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#### Zeta potential measurement

Aqueous suspensions of brookite TiO<sub>2</sub> nanorods (1.0 mg/mL) with various pH values were <sup>18</sup> used for zeta potential measurements. The pH values of the aqueous suspensions were adjusted with hydrochloric acid by monitoring with a pH meter. Thus, the salt concentrations <sup>20</sup> of the suspensions depend on the pH conditions. For example, the estimated concentration <sup>21</sup> of hydrochloric acid at pH 2 was lower than 10 mM. Measurements were performed with <sup>22</sup> a Zetasizer Nano ZS instrument (Malvern Instruments, UK). The aqueous suspensions were <sup>23</sup> sonicated for 1 min before the zeta potential measurements were performed. <sup>24</sup>

#### AFM sample preparation by electrostatic method

TiO<sub>2</sub> nanorods were fixed by an electrostatic method as follows. We prepared aqueous <sup>2</sup> suspensions (1.0 mg/mL) of TiO<sub>2</sub> with the same condition of zeta potential measurements. <sup>3</sup> An suspension of TiO<sub>2</sub> (200  $\mu$ L) was sonicated for 1 min and deposited on a freshly cleaved <sup>4</sup> mica surface (a round disc of diameter 12 mm, purchased from SPI Supplies). The sample <sup>5</sup> was incubated for 1 h at room temperature and rinsed with KCl aqueous solution (100 mM). <sup>6</sup>

#### AFM sample preparation by cross-linking method

TiO<sub>2</sub> nanorods were fixed by cross-linking with a phosphate-terminated self-assembled mono-8 layer (SAM). A Au(111) substrate was prepared according to a previously reported proce-9 dure.<sup>53</sup> After opening the vacuum chamber to the atmosphere, the Au(111) substrate was 10 immediately immersed in an ethanol solution of 12-mercaptododecylphosphonic acid (745855, 11 Sigma Aldrich) to form a SAM, and left overnight in the dark. The phosphate-SAM sub-12 strate was rinsed with ethanol and pure water and then dried with  $N_2$  gas. A suspension 13 of brookite  $TiO_2$  nanorods (2.0 mg/mL) was prepared with ultra-pure ethanol (Wako Pure 14 Chemical Industries, Ltd.). After sonication for a 1 min, the suspension (50  $\mu$ L) was dropped 15 on a substrate with the phosphate-SAM (diameter: 12 mm) and left to stand for 1 h in a 16 sealed vessel to suppress evaporation. After removal of unadsorbed nanorods by rinsing with 17 ethanol, the substrate was heated at 120°C for 24 h and cooled to room temperature. The 18 substrate was rinsed with a KCl aqueous solution (100 mM) before the AFM measurements. 19

#### FM-AFM measurement in liquid

The AFM measurements were performed with a laboratory-built ultralow-noise deflection <sup>21</sup> sensor as previously reported.<sup>41–43</sup> The laboratory-built AFM apparatus was combined with <sup>22</sup> a commercially available AFM controller (Nanonis RC-4, SPECS Zurich GmbH, Zurich, <sup>23</sup> Switzerland). All AFM investigation of the brookite TiO<sub>2</sub> nanorods were performed at room <sup>24</sup>

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temperature in 0.1 M KCl aqueous solution. The high salt concentration (0.1 M) can help us 1 to obtain subnanometer-scale resolution in the FM-AFM imaging by suppressing the long-2 range tip-sample interaction force. We can enhance the relative contribution of the short-3 range tip-sample interaction force to the contrast formation in the obtained FM-AFM images. 4 A commercially available silicon cantilever (PPP-NCHAuD, Nanoworld, Switzerland) with 5 a nominal spring constant of 42 N/m and a resonance frequency of 150 kHz in a liquid 6 was used. A tip side of the cantilever was coated with Si (thickness: 30 nm) by dc sputter 7 coater (K575XD, Emitech) before each AFM measurement to improve the reproducibility of 8 atomic-resolution AFM imaging in liquid as previously reported.<sup>54</sup> 9

#### DFT MD simulation

Accurate modeling of the solid-liquid interface between brookite and water, and hence the hydration structure of the crystal, was achieved by dispersion-corrected DFT calculations. Exchange-correlation effects were described by the generalized gradient approximation level PBE<sup>55</sup> functional augmented by the Grimme D3 dispersion correction. <sup>56</sup> All DFT calculations were performed using the CP2K package. <sup>57</sup> Further details of the simulation methods are available in a previous report. <sup>52</sup>

## **Results and discussion**

The brookite  $TiO_2$  nanocrystals used in this study were rod shaped with their {210} and {212} surfaces exposed, as shown in Figure 1a. We synthesized the brookite  $TiO_2$  nanorods with the procedure that we previously reported. <sup>51,52</sup> Figure 1b shows the atomic-scale surface structures of the (210) brookite  $TiO_2$  surface. In this study, we propose two methods to fix the brookite  $TiO_2$  nanorods on a substrate for atomic-resolution FM-AFM measurements. 22

The first fixing method is based on the electrostatic attraction between brookite  $TiO_2$  <sup>23</sup> nanorods and muscovite mica. The cleaved surface of muscovite mica, which is widely <sup>24</sup>

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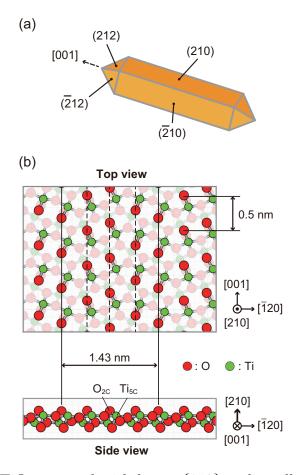


Figure 1: (a) Brookite  $TiO_2$  nanorod with larger {210} and smaller {212} exposed crystal faces. (b) Atomic-scale models of brookite (210) surface (top and side views). Red and green colors indicate oxygen (O) and titanium (Ti) atoms, respectivelly. In the top view model, the topmost atoms are enhanced by depth fading display. The size of a repeating unit of atomic rows along [001] direction is 1.43 nm. Three additional atomic rows exsist in the single repeating unit as indicated by dashed lines.

used as a substrate for AFM measurements, has a negative charge at a wide range of pH 1 values. A previous study reported that a mica surface is negatively charged even under 2 pH 2 condition,  $^{58}$  suggesting that the isoelectric point of a mica surface is lower than pH 3 2. If the brookite  $TiO_2$  nanorods have positively charged surfaces, there is a electrostatic 4 attraction (coulombic attraction) between the brookite  $TiO_2$  nanorods and the mica as shown 5 in Figure 2a. Zeta potential measurements were performed to investigate the surface charges 6 of the brookite  $TiO_2$  nanorods under different pH condition (Figure 2b). The results suggest 7 that the brookite  $TiO_2$  nanorods have positively charged surfaces under acidic conditions 8

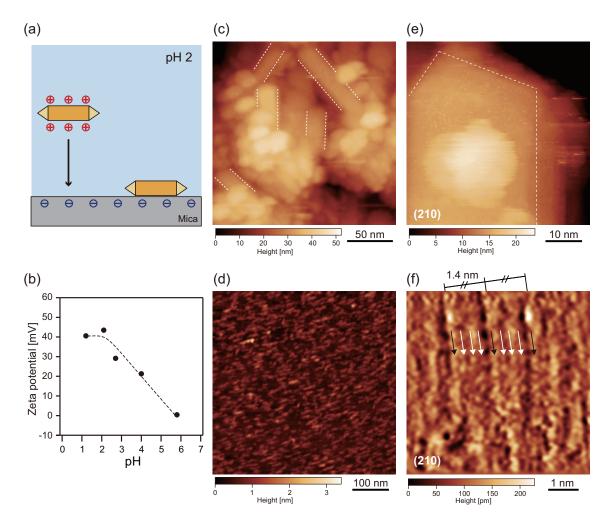


Figure 2: (a) Diagram of adsorption of  $\text{TiO}_2$  nanorods on mica substrate at pH 2 by electrostatic method. (b) Zeta potentials of  $\text{TiO}_2$  nanorods at various pH values. FM-AFM images of brookite  $\text{TiO}_2$  nanorods prepared with (c) pH 2 and (d) pH 5 aqueous solutions. (e) FM-AFM image taken near end of brookite  $\text{TiO}_2$  nanorod. (f) Atomic-resolution FM-AFM image of brookite (210) surface. Periodic contrasts with the spacing of 1.4 nm along [001] direction showed good agreement with the repeaing unit of brookite  $\text{TiO}_2$  crystal (see Figure 1b). White arrows indicate subnanometer-scale structures between the periodic contrasts of 1.4 nm spacing that are visible only at the very limited area of the image. The dashed lines in Figures 2c and 2e indicate the crystal shapes of brookite  $\text{TiO}_2$  nanorods. All FM-AFM images were obtained in 0.1 M KCl aqueous solution.

(pH less than 6); the zeta potential increases with decreasing pH, and reaches saturation at 1 around pH 2. On the basis of the zeta potential measurements, an acidic aqueous suspension 2 (pH 2) of brookite TiO<sub>2</sub> nanorods was used, and was dropped onto the mica surface. As 3 shown in Figure 2c, the adsorbed brookite  $TiO_2$  nanorods prepared in an acidic aqueous 4 suspension (pH 2) were stable during FM-AFM imaging in 0.1 M KCl aqueous solution. In 5 contrast, the brookite  $TiO_2$  nanorods deposited from a neutral aqueous suspension (pH 5) 6 were not fixed on the substrate (see Figure 2d). The results show that the pH conditions are 7 related to the adsorption step of TiO<sub>2</sub> nanorods onto the mica surface because the aqueous 8 suspensions were only used for the adsorption step in this study. It is generally possible that q the high salt concentration weakens electrostatic interactions between the brookite  $TiO_2$ 10 nanorods and mica. However, the adsorption structures of brookite TiO<sub>2</sub> nanorods that 11 are prepared at pH 2 were stable even after the change of the solution from the adsorption 12 solution (HCl, pH 2, <0.01 M) to the rinse solution (KCl, 0.1 M). We demonstrated that 13 brookite  $TiO_2$  nanorods in an acidic suspension (pH 2 in this study) can be fixed on mica 14 via electrostatic attraction. 15

Figure 2e shows an FM-AFM image taken near the end of a brookite TiO<sub>2</sub> nanorod fixed 16 by the electrostatic method. The characteristic shapes of the brookite  $TiO_2$  nanorods were 17 visualized in the AFM image, as indicated by the dotted lines in Figure 2e. The shapes in the 18 AFM image agreed with those previously obtained by transmission electron microscopy <sup>51</sup>. 19 We therefore assigned the crystal orientation of the brookite  $TiO_2$  nanorods from the direc-20 tions of the crystal shapes. As an initial step in the AFM imaging, we looked for fine brookite 21  $TiO_2$  particles showing a rod shape and a flat side plane of (210) surfaces (see Figure S3 in 22 Supporting Information). After that, a tip was brought to the flat area of brookite (210) 23 surface and scanned with a reduced scan range. In this way, we acquired successive FM-AFM 24 images in the order of Figure S3, Figure 2e, and Figure 2f to assign brookite (210) surfaces. 25 An FM-AFM image of the (210) surface (Figure 2f) shows subnanometer-scale contrasts 26 with striped structures parallel to the |001| direction. The periodic spacing of the striped 27

structures (around 0.3 nm) shows good agreement with the distance between atomic rows  $_{1}$  of the topmost oxygen or titanium atoms (see Figure 1b). Here, we found slightly higher  $_{2}$  striped structures, with a periodicity of one out of four rows, as indicated by the arrows in  $_{3}$  Figure 2f. The origin of the slightly higher striped structures at the (210) brookite surface  $_{4}$  cannot be explained by the crystal structure of brookite TiO<sub>2</sub> alone. We will discuss the  $_{5}$  possible origin later.

The FM-AFM images show atomically flat terraces and atomic-scale contrasts at the  $^{7}$  surface. In general, atomic-resolution AFM imaging of particle-shaped samples is very challenging because of the difficulties associated with particle fixation. There are therefore only  $^{9}$  a few reports of atomic-resolution AFM imaging of particle-shaped samples in a liquid.  $^{50}$   $^{10}$  The results of this study show that an electrostatic fixing method can be useed for atomicresolution AFM imaging of the TiO<sub>2</sub> nanoparticles in a liquid.  $^{12}$ 

Covalent cross-linking is the second method used for fixing brookite  $TiO_2$  nanorods for 13 AFM measurements. The brookite  $TiO_2$  nanorods were fixed with a SAM of a phosphate-14 terminated alkanethiol on a Au (111) substrate. Phosphate groups form covalent bonds 15 with various metal oxides, including  $TiO_2$ , by a dehydration reaction. In this study, an 16 alkanethiol SAM of 12-mercaptododecylphosphonic acid on Au (111) was used (Figure 3a) 17 as a substrate. After spin-coating with a suspension of brookite  $TiO_2$  nanorods, the substrate 18 was heated at 120 °C for 24 h to accelerate the dehydration reaction between phosphate and 19 the TiO<sub>2</sub> surface. Figure 3b shows an FM-AFM image of a brookite TiO<sub>2</sub> nanorod, that was 20 fixed on the phosphate-terminated SAM by covalent cross-linking. The nanorod shape of the 21 brookite  $TiO_2$  is visualized in the FM-AFM image. An atomic-resolution FM-AFM image 22 of the brookite  $TiO_2$  nanorods (Figure 3c) was obtained at the center of the nanorod shown 23 in Figure 3b. The image (Figure 3c) shows rows consisting of periodic dot-like contrasts 24 aligned in the [001] direction. As shown in Figure 3d, the periodic spacing of the dots in 25 the [001] direction is 0.5 nm, which is consistent with the expected structures of the (210) 26 brookite surface (see Figure 1b). 27

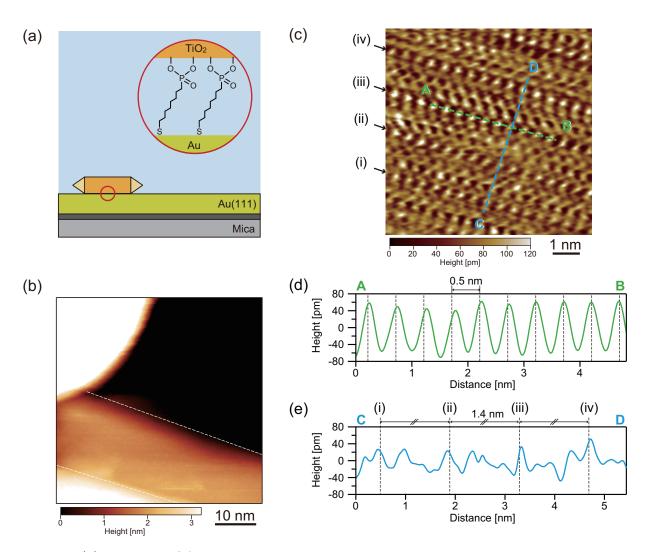


Figure 3: (a) Diagram of AFM sample preparation by cross-linking method. The inset shows the chemical cross-linking structures of phosphate-terminated alkanethiols between Au(111) substrate and brookite  $TiO_2$  nanorods. (b) FM-AFM image of brookite  $TiO_2$  nanorod. The dashed lines indicate the crystal shape of a brookite  $TiO_2$  nanorod. (c) Atomic-resolution FM-AFM image of brookite (210) surface. (d) and (e) are height profiles along line AB and line CD in (c), respectively. All FM-AFM images were obtained in 0.1 M KCl aqueous solution.

The quality of the AFM image (Figure 2f) obtained with the electrostatic method is 1 lower than that obtained with the cross-linking method (Figure 3c). Here, we discuss a 2 possible reason for the difference in the quality of subnanometer-scale AFM images between 3 the two fixing methods. In general, conditions of tip apex affect the quality of atomic-4 scale FM-AFM images. The estimated radius of the Si coated tips used in this study was 5 approximately 30 nm. This tip radius is much larger than an atomic scale. However, even 6 with such a nanoscale tip, we are able to obtain atomic-scale FM-AFM images. The details of 7 the imaging mechanism were previously reported.<sup>45</sup> In short, subnanometer-scale contrasts in 8 the FM-AFM images obtained in liquid originates mainly from the free energy change caused q by the events happening at the space between the tip front atom and the surface topmost 10 atom. Therefore, atomic-scale FM-AFM images can be obtained even with a relatively 11 blunt tip having a nanoscale radius. This discussion suggests that the detailed patterns of 12 the atomic-scale contrasts in the FM-AFM images are certainly affected by the structure 13 and stability of the atomic-scale protrusion and the hydration structure formed under it. In 14 our experiment with the samples prepared with the electrostatic method, it was difficult to 15 find atomically flat surfaces of brookite (210) due to the large surface corrugation as shown 16 in Figure 2c. Thus, we speculate that the tip apex was contaminated or damaged during 17 the tip scanning for finding the flat (210) surfaces. In contrast, in the experiment with the 18 samples prepared with the cross-linking method, it was relatively easy to find the atomically 19 flat surfaces of brookite (210) owing to the small surface corrugations (see Figure S4 in 20 Supporting Information). 21

We found that some of the characteristic rows show a brighter contrast as indicated <sup>22</sup> by arrows (i)-(iv) in Figure 3c. Although the regularity of the contrast is not perfect, the <sup>23</sup> average distance between the rows with a brighter contrast is 1.4 nm (Figure 3e). Such <sup>24</sup> brighter rows with a spacing of 1.4 nm are also found in the FM-AFM image of the brookite <sup>25</sup> TiO<sub>2</sub> nanorods fixed by the electrostatic method (Figure 2b). This agreement suggests that <sup>26</sup> the bright rows correspond to the same structural feature. <sup>27</sup>

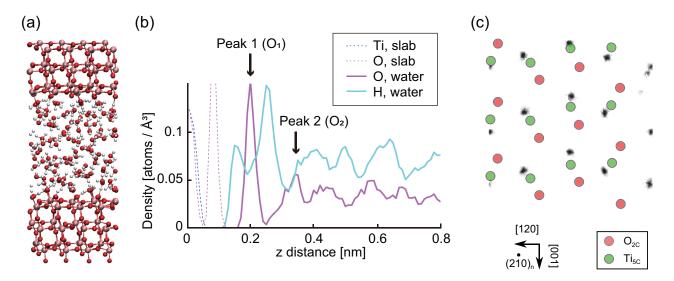


Figure 4: (a) Snap shot of atomistic model used for DFT MD simulation of brookite TiO<sub>2</sub> (210)-water interface. (b) Vertical density profiles of oxygen (O) and titanium (Ti) atoms of brookite structures (slab) and oxygen (O) of water molecules. The two peaks of water molecules are defined as the first hydration structure (O<sub>1</sub>) and second hydration structure (O<sub>2</sub>), from the closest to the surface. (c) Lateral positions of topmost atoms and water molecules of first hydration layer. The darker spots represent the histogram of lateral positions of oxygens of water molecules within the first oxygen density peaks (O<sub>1</sub>) of (b).

The contrast formation mechanism in the subnanometer resolution FM-AFM images was 1 investigated by performing DFT MD simulations of a brookite-water interface. Previous 2 reports have indicated that not only the outermost atoms, but also the hydration structures 3 at the solid-liquid interface contribute to contrast formation in the subnanoscale FM-AFM images in water. Recently, Holmström et al. used the DFT MD simulations of brookite 5 TiO<sub>2</sub>-water interfaces to clarify the hydration structures and formation of OH-terminated 6 structures at the surfaces in aqueous solutions.<sup>52</sup> In this study, the results of the FM-AFM 7 experiments and DFT MD simulations were compared to determine the origin of the subnanoscale contrasts in the AFM images in water of brookite  $TiO_2$  (210) surfaces. Figure 4a 9 shows a snapshot of the atomistic model used for the DFT MD simulations. The simulations 10 predict the spatial distribution of water molecules at the surface of brookite  $TiO_2$  (210). 11 Two clear peaks appear in the vertical density profile of oxygens, as shown in Figure 4b. 12 The results suggest that structured water molecules are present on the brookite  $TiO_2$  (210) 13

surface. In this study, the two peaks are defined as the first hydration structure  $(O_1)$  and 1 second hydration structure  $(O_2)$ , from the closest to the surface. Figure 4c shows the lateral 2 positions of oxygen  $(O_{2C})$ , titanium  $(Ti_{5C})$  and water molecules. The black distributions in 3 Figure 4c indicate the lateral distributions of the oxygens of peak 1  $(O_1)$ . The simulation 4 suggests that most of the water molecules in peak 1 were adsorbed on the surface, with 5 formation of a chemical bond between a water oxygen and a surface  $Ti_{5C}$  ion. In addition, 6 water molecules form hydrogen bonds with surface oxygens. The adsorption energy of a 7 water molecule on brookite  $TiO_2$  (210) is 1.1 eV. As a result of stable interactions, lateral 8 ordering of water molecules is observed in the first hydration layer (Figure 4c). In our preq vious study, $^{52}$  we have investigated the influence of ions (potassium and chloride ions) on 10 hydration structures at the brookite-water interfaces by DFT MD simulation. According to 11 this theoretical study, the lateral distribution of time-averaged hydration layers are not sig-12 nificantly affected by the existence of ions. In the meanwhile, experimental results reported 13 from different research groups demonstrated that atomic-scale contrasts in the 2D FM-AFM 14 images of various surfaces obtained in an electrolytic solution can be largely explained by 15 the arrangement of the surface topmost atoms and/or the hydration structures even without 16 taking into account the influence of ions.<sup>45,59–62</sup> 17

As mentioned above, to clarify the origin of the subnanoscale contrasts obtained in wa-18 ter, the contributions from the topmost atoms ( $Ti_{5C}$  and  $O_{2C}$ ) and the hydration structures 19 should be taken into account. The bright particle-like contrasts therefore probably arise 20 from the first hydration structures or the topmost oxygen atoms  $(O_{2C})$ . Two possibilities 21 (oxygen atoms or first hydration structures) remain as possible origins of the subnanoscale 22 contrast. In addition, a unique assignment of the [001] direction in the FM-AFM image is 23 impossible on the basis of only the obtained data. Four possible models therefore remain for 24 assignment of the subnanoscale contrasts to the surface structures, as shown in Figure 5. In 25 models I and III, the positions of the oxygen atoms and the bright particle-like contrasts are 26 superimposed. Models II and IV show superimposed images of water molecules in the first 27

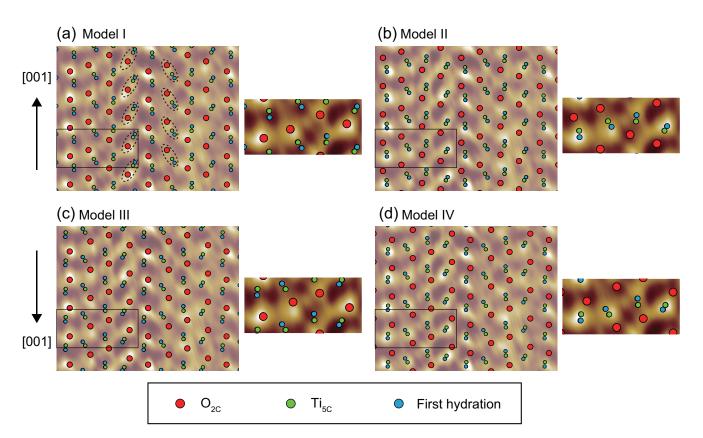


Figure 5: Four possible arrangements of topmost atoms and water molecules of first hydration layer for atomic-scale contrast in FM-AFM image. Positions of  $O_{2C}$  [(a) and (c)] and hydration structures [(b) and (d)] were adjusted to give bright contrasts of AFM image. Arrows with [001] indicate the directions of topmost atoms of brookite and hydration structures. Dashed lines in (a) indicate tilted contrasts in the AFM image.

hydration structures and the bright contrasts. In some atomic rows, the tilted contrasts in 1 the FM-AFM image agree well with the arrangement of adjacent Ti-O-hydration pairs, as 2 indicated by the dashed lines. Although it is difficult to assign a definite arrangement from 3 the four possible candidates, the bright particle-like contrasts are more likely to originate 4 from the first hydration structures. In a previous FM-AFM investigations of calcite in wa-5 ter,<sup>63</sup> subnanoscale contrasts were formed by scanning a tip on the first hydration structures, 6 because of the high adsorption energy of water molecules on calcite (approximately 1 eV). 7 The adsorption energy of water molecules on brookite  $TiO_2$  (210) (1.1 eV) is comparable to 8 that on calcite. These results suggest that the tip is scanned on the first hydration structures c when subnanoscale contrasts of brookite  $TiO_2$  (210) are formed. 10

For the bright rows, we were unable to determine the contrast formation mechanism 11 even from the results of the DFT MD simulation. One possible model is OH termination at 12 the brookite (210) surface by dissociative adsorption of water. Although adsorption of water 13 molecules in the dissociated state was observed in the DFT MD simulation, it was impossible 14 to obtain statistical data to investigate this as possible origin of the bright rows observed in 15 the FM-AFM images because of the huge computational costs. As another possible origin 16 of the bright rows, it may be caused by the influence of ions. However, it is very difficult 17 to quantitatively estimate the influence of ions with present our results. Although further 18 studies are needed to determine the origin, the atomic-scale FM-AFM images suggest the 19 presence of characteristic structures with a regularity of one in four rows at the brookite 20 (210) surfaces. 21

## Conclusions

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In this study, we established two fixing methods to enable the AFM investigations of brookite  $^{23}$ TiO<sub>2</sub> nanoparticles in a liquid. The two fixing methods were used for visualization of atomic-  $^{24}$ scale surface structures of brookite (210). The visualization mechanisms were investigated  $^{25}$  by calculating the distributions of the topmost atoms and hydration water molecules at the 1 interface by DFT MD simulations. Comparisons of the subnanoscale contrasts in the FM-2 AFM images and the distribution of the topmost atoms and water molecules showed that the 3 subnanoscale contrasts originated from the first hydration structures or the topmost oxygen 4 atoms. Moreover, although the origin was not completely clear, we found the presence 5 of bright rows of particle-like contrasts with a periodicity of one out of four rows at the 6 interface between brookite (210) and water. The established fixing methods help to improve 7 our understanding of the photocatalytic reactions of TiO<sub>2</sub>, not only brookite, but also rutile 8 and anatase. 9

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## Supporting Information Available

The following files are available free of charge. SEM image and XRD patterns of the  $_{21}$  prepared brookite TiO<sub>2</sub> nanorods. AFM image of a sample prepared with the cross-linking  $_{22}$  method.  $_{23}$ 

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## Graphical TOC Entry

