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Mechanical properties of 3-D printed truss-like lattice biopolymer non-stochastic structures for sandwich panels with natural fibre composite skins


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Abstract

A full mechanical characterization of three types of 3-D printed lattice cores was performed to evaluate the feasibility of using additive manufacturing (AM) of lightweight polymer-based sandwich panels for structural applications. Effects of the shape of three selected lattice structures on the compression, shear and bending strength has been experimentally investigated. The specimens tested were manufactured with an open source fused filament based 3-D printer. These sandwich structures considered had skins made of polypropylene (PP) flax bonded to the polylactic acid (PLA) lattice core using bi-component epoxy adhesive. The PP flax and the PLA core structures were tested separately as well as bonded together to evaluate the structural performance as sandwich panels. The compression tests were carried out to assess the in-plane and out-of-plane stiffness and strength by selecting a representative number of cells. Shear bands and plastic hinges were observed during the in-plane tests. The shear and three-point bending tests were performed according to the standard to ensure repeatability. The work has provided an insight into the failure modes of the different shapes, and the force-displacement history curves were linked to the progressive failure mechanisms experienced by the structures. Overall, the results of the three truss-like lattice biopolymer non-stochastic structures investigated have indicated that they are well suited to be used for potential impact applications because of their high in-plane and out-of-plane compression strength. These results demonstrate the feasibility of AM technology in manufacturing lightweight polymer-based sandwich panels for potential structural uses.
1. Introduction

Sandwich structures are advantageous for structural applications exhibiting low-weight, high-stiffness and high energy absorption [1-3]. Depending on the industry standards and functional requirements, sandwich cores come in different forms and materials including honeycomb and metals for aerospace [4, 5] and marine applications [6], natural composites [7] foam, balsa, and wood for automotive [2] and many other applications including construction [8]. Sandwich panels consist of two thin face skins contributing to the flexural stiffness of the panel, whereas the core is responsible for the shear transfer between the skins. The core can also contribute to the overall flexural and compressive strength of the panel. Cellular sandwich structures have been developed for various structural requirements in order to develop light and adequately stiff and strong composites. Advancement in additive manufacturing (AM) technology has led to the emergence of metallic cores in the form of cellular honeycomb structures and truss-like structures specifically for high-performance mechanical properties typically used in lightweight energy absorbing products [9-12]. Much attention and development activities have been dedicated to produce optimized metallic structures through improved designs and build parameters [13]. There is a potential to use cellular honeycomb structures in polymer applications as well, but reinforced polymers and polymer composites would be preferred for improved mechanical strength. A comprehensive study by Yazdani et al. [14] covered the theoretical, experimental, and practical aspects of cellular and truss-like structures. The core is a potential to use cellular structures in truss-like polymers [14] to improve the core's mechanical properties. A comprehensive study by Yazdani et al. [14] covered the theoretical, experimental, and practical aspects of cellular and truss-like structures.

Numerical and experimental aspects of the geometry optimization of 3D printed cellular cores for energy absorption applications. Bio-composites offer the promise to improve sustainability and functionality compared to oil-derived polymer materials. 3-D printing (3DP) these materials for future applications because of geometric flexibility (e.g., honeycomb lightweight structures) is of particular interest [15]. Natural fibre (NF)-based composites have recently attracted interest due to increased ecological concerns in global industry since issues of manufacturing-based energy burdens, recyclability and environmental safety become the centre of new materials and product developments [16, 17]. Among these natural fibres, lignocellulosic bast fibres such as flax, hemp and jute are widely used for reinforcement of thermoplastic polymers. The reinforcement of renewable sourced polymers with flax fibres is encouraged due to their low density, good mechanical properties and low environmental impact. Flax and polypropylene (PP) has been shown to be compatible for making high-quality composites [18]. Although PP can be printed, it is more challenging than the natural polymer poly(lactic acid) (PLA), which is the most common fused filament fabrication (FFF)/fused deposition modeling (FDM) 3-D printed polymer due to its low melting temperature and ease of printing PLA can be used by consumers to make finished products [19]. The mechanical properties of FFF/FDM PLA are well established [20-23], and their chemical resistance is also known [24]. PLA has also been successfully used in a number of polymer matrix composites [25-30].

Biocomposites offer the promise to improve sustainability and functionality compared to oil-derived polymer materials. 3-D printing (3DP) these materials for future applications because of geometric flexibility (e.g., honeycomb lightweight structures) is of particular interest [15]. Natural fibre (NF)-based composites have recently attracted interest due to increased ecological concerns in global industry since issues of manufacturing-based energy burdens, recyclability and environmental safety become the centre of new materials and product developments [16, 17]. Among these natural fibres, lignocellulosic bast fibres such as flax, hemp and jute are widely used for reinforcement of thermoplastic polymers. The reinforcement of renewable sourced polymers with flax fibres is encouraged due to their low density, good mechanical properties and low environmental impact. Flax and polypropylene (PP) has been shown to be compatible for making high-quality composites [18]. Although PP can be printed, it is more challenging than the natural polymer poly(lactic acid) (PLA), which is the most common fused filament fabrication (FFF)/fused deposition modeling (FDM) 3-D printed polymer due to its low melting temperature and ease of printing PLA can be used by consumers to make finished products [19]. The mechanical properties of FFF/FDM PLA are well established [20-23], and their chemical resistance is also known [24]. PLA has also been successfully used in a number of polymer matrix composites [25-30].

These advances in sandwich structures and polymer/natural fibre composites open up the potential to overcome one of the primary limitations of FFF/FDM in load bearing applications: highly anisotropic properties having particularly low stiffness and strength in the direction of layer deposition (z-direction) [31-34]. A review by Ngo et al. [35] summarized the main methods commonly utilized in additive manufacturing among which FDM was discussed. In FFF/FDM three-dimensional parts directly imported from CAD models are fabricated based on the deposition of melted filaments to create an object as a composition of several overlapping layers, where each layer is composed of a continuous bead extruded and deposited which solidifies by cooling [31]. Among the complex geometries that can be produced using FFF/FDM is the lattice or cellular structure for load-bearing applications, and it is expected that the mechanical properties of the sandwich cellular structures for load-bearing applications could be improved by fibre reinforced thermoplastics.

In order to probe these potentials, a study was designed to take advantage of the more advanced work in PP-flax composites to make up the top and bottom layers of the sandwich structure and PLA to form 3-D printed truss-like lattice units making up the internal ones. The aim of this work thus was to collect sufficient mechanical properties of three selected lattice topologies of PLA and PP+flax composites to be used in the structure optimization process for sandwich structures. Mechanical tests included compression both in-plane and out-of-plane (edgewise and flatwise), core shear and sandwich flexural. Force vs displacement curves were generated, which were then processed...

ac cor din gly t o i nt er pr e t the a ch i ev ed  r es ul t s  an d l a t er  di s cu s s ed  t he  ov er al  l r es pon s e  t o  dr aw
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The po t en t i a l  f or  FFF/FDM t o  pr ov i de  l oad - b e a r i n g  s t r uct ur es  f or  v a r i ou s  a p pl i cat i on s  wi t h
thes e  m at er i a l s  ar e di s cu s s ed ,  an d  con cl us i on s  ar e  dr aw n.

2. Mate r i a l s  and m et hodol ogy

2.1. Lattice s tr uct ur e s

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sto  c alt i c  a  e  ca  e  g  s er da  c odi ng  t o  t he  r i  b u i l d i ng  u n i t  c  l i e t h e  r 2  D  la t i c  e u ni t
(ho n  y mb) o  r 3-D la t i c  e u ni t  (truss-like ) [3 6 ] . T he  l a t t i c e  s tr uct ur e  i s  a  tr uss - l i k e
s tr uct ur e  w i t h  a  r e  p  a i n g  u n i t  c e l l  t h a t  fo m s  i nt e  s e  i n g  str u t s  a  r  i  no  d  i  s  w i t h  a  s p e  d i  c
r e  c r i ng  a  m g e  m n  o  v  m  vo  u me  f i c  r e  g  n  . T he  i nt e  i d le  g  n  f i e  s  s tr u t s  w h e  r  e  r
ho lb w  c i r c  l a  r  sq u a  e  o  r a  y  d e  i r e  d s h a  p c  a  s  s e  d o t  a  i s  a  p l i c  t i o n  de  p  de  n  a  r
m o  e  s p e  i f i c  d y l d e  p  de  n  o  n t h e  s tr uct  h a  a  r  i  st i f n e  s  r e  q u i r e  m  p s [3 7 ] . K a  gm c  trus
s e  n  a  r  str uct u re  s e  fo m  be  t e  n h a  n  o v n  t i o n  m  h o  n  y  o  b m  c  str uct u re  s i n  te  m s  o f  t het
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s tre  g  h  o f  t he  k  g  a e  s truct ur e , t he  i nt e  m a  m  c  a  h e  a  d u t e  d c  d  e  r t o  4 5  o  r 6 0
re  m  e  d v  e  y . T he  thre  e u ni t  c e l l  s tr uct ur e  s w e  a  de  e lo  p  d w it h  t he  CAD  s o  t h w a  e  CATIA
V5 (Dassault Systèmes, Velizy-Villacoublay, Fra n c e ) m o  n a te  d  f r o m  t he  p re  i o n y
inve t i g  a e t s tr uct ur e  s i n  lin e  a t u re . Ho w  e  m o  d i c  t i o n  t o  t he  g e  e n t y  a  r  d  d i m e  s i o  n

have been applied to improve the print quality a die ema turing time s. Latic e structure s are e g ee by re pting unit c ils, whic in c ndic struts built a dffe o o c on te dhro y hno dl sFo xo pa iso npurpo e sa fixe or ed po fidht a d heig ho fl 0x1 0mm is use d. The STL file sof the de a ca ne fo nd at the Ope Frame wk [3] a d a sho w in Figure 1.

Figure 1 - a) Lattice 1: BCZ structure b) Lattice 2: circular rectangular pattern of 4 vertical struts c) Lattice 3: BCZ unit cell without the vertical strut.

Lattice 1 is a diamond design that has a vertical strut that runs through the middle of the unit cell. The vertical strut in the c etme o fthe c h has a dia mte o 2.5 mm whilst the remaining dia g al struts ha v a re dc e thic k s o fl 5 mm. With the ve tic h strut pre s n in the la v t of the unit c d, lthe c mpre sive c aaplitie so fthe c h sho il se e g e te simpro v m n co pa e do the sa m de ig mitho u. Fo the se c d unit c d, k c be - like fom a tha see untlie d with five ve tic struts (4 o rea c d cor with 1 in the ce tme ) a d e g h smalle struts that me eat the middle of the ve tic hstrut in the c etme o fthe c h. Simila rto unit c d 1 , the ve tic struts ha v a thic k s o f2.5 mm whe a th the smalle r dia g al strut be a thic k s o fl 5 mm. This c ell struc ture is e pe te dto pe fo m e pe in ly well unde mpre sio ndue to a nine c e sin ma s a sa re ult of the a ditio n l

Fig we 1 - a La tic e: B (C-Z struc t ure b) La tic e 2 rirc mfe or rial re a tr y ia pa te u of 4 ve tic struts a dc c La tic e: B CC un it c o witho u the ve tic astrut.

Vertical struts in comparison to unit cell 1. The third unit cell to be observed is a cell that also takes a diamond-like that of unit cell 1. However, it differs in the fact that it does not have the vertical strut present in the centre of the cell. As a result, it is expected to perform to a lower standard to unit cell 1 under compression. Due to the absence of vertical struts in this cell, the thickness of the struts is marginally thicker than the other two cells, with a diameter of 2 mm instead of the original 1.5 mm.

2.2. Manufacturing

Manufacturing of specimens was carried out using an open source Ultimaker 3 Extended dual extruder (Ultimaker, the Netherlands) equipped with 0.4 mm nozzles on a 215 x 215 mm heated glass build plate. CAD files were imported into Cura 2.7.0 [39] in STL format to be sliced, and a list of printing parameters was adjusted accordingly. These instructions are subsequently generated in a G-code script, which the printer uses to perform the printing. A process of fine-tuning the printing parameters was to a thin a no timum quality where a visual inspection of printed parts was carried out to identify physical defects, which could potentially yield inconsistent results following [40]. No support material was used for all printing operations in order to make this method of manufacturing realistic in mass production. In addition, printing without support reduces energy and material, which reduces costs as well as environmental impact.

The complete list of parameters used to 3-D print all types of specimens is shown in Table 1.
La 3-D printing par a m a s use do ma nfa cure the sa iple st e te  dn this wo k.

<table>
<thead>
<tr>
<th>Layer Thickness (mm)</th>
<th>0.1</th>
</tr>
</thead>
<tbody>
<tr>
<td>Infill density (%)</td>
<td>100</td>
</tr>
<tr>
<td>Print temperature (°C)</td>
<td>225</td>
</tr>
<tr>
<td>Build bed temperature (°C)</td>
<td>60</td>
</tr>
<tr>
<td>Print speed (mm/s)</td>
<td>15</td>
</tr>
<tr>
<td>Flow rate (%)</td>
<td>100</td>
</tr>
<tr>
<td>Build bed glue stick</td>
<td>Glue stick</td>
</tr>
</tbody>
</table>

2.3 Materials

The feedstock for the 3-D printed cores used in this study was a PLA filament made of starch promoting its bio-gradability and biocompatibility. Spool of blue PLA 750 g of 2.75 mm diameter, melting temperature range of 145-160 °C and a specific density of 1.24 purchased from RS Components Ltd, UK. The mechanical properties of bulk PLA were reported in the datasheet [41] with a tensile modulus of 2852 MPa and yield stress of 38.5 MPa, which were tested in accordance with ISO 527.

Natural fibre (NFC) face sheet made of discontinuous flax fibres and polypropylene and preformed with a mass content ratio 52:48 flax/PP were produced at EcoTecnilin Ltd, Cambridge, UK. These consolidated mats called fibrilastics sheets are a mixture of discontinuous flax fibres and PP interlocked using needle punch process with 30,000 needles/s resulting in the areal weight of 1200 gs/m. Consolidation using heated press with applied pressure of 20 MPa and 120 °C for 5 minutes yielded a reduced thickness of 2 mm. The motivation for the use of NFC is to reduce carbon footprint.
Furthe mo e pro ple a is a fo sil fuel ba s dt he mo pa sic with a ma inum te sile mo dius of 1.5 GP aa d stre g h of 3 (MP a[4]

Fo the sa ndwi hp a nl's pro dc io u the a de is no fhe 3 D printe do the NF fa ce she es twa sa cih dusing Araldite 2 0 1(AV5 3 0 $IV5 3 0l) bi-c mpo n h e p x pa te a de ise purc h e drom R S Co mpo n ts Ltd, N otha ts, UK. The two pa te p x re in a de ha de n rwe x mixe din a ne qa la m uts a d a lb w dto re a tfo r5 minute sin ro o m te me x tre befo e being a plie do the fa c she es to ly.

2.4 Me hanical te sigh

2.4.1 Tensile testing o fNF C

The si le pro p tie s (i.e., stiffne s a d stre g h) o f NFC we e x a din a c odi n e with ISO 5 2 4- 9 a7id sta dia te ts spe cmic type 1 Bwith no e di ta bi ng using the e d crot me h ic a te sing Tinius Olse n2 5ST with 2 5kN lo a di (Figure 2).

Te ts we e pe fo me do nspe cmic nof 1 5 0 2 0x 2 mm in dime si o na ta cro she a d displac e cmio 2 mm/min a d g agude gh of 5 0mm of a re dc e wth o fl 0mm. Te ts we e c ar do a tro o me ape x tre a d a tie at three re ptitio n we e pe fo me d a di ra we si ts we e a x g etola me a valu e with Tinius Olse n1Ho iz o so fwa e.
2.4.2 Compression testing of core lattice structures

Quasi-static compression tests in-plane and out-of-plane were performed on the three core lattice structures using the same Tinius Olsen 25 ST machine. In-plane tests are also called geometric compression and are used to determine the compressive strength and stiffness in the direction parallel to the sandwich facing plane. Specimen tested were only lattice cores; therefore, buckling was insignificant. These tests were constructed to replicate the ASTM C364-99 standard as closely as possible with some modifications to the geometry for clamping purposes. Dimensions of the cubic tested specimen were dependent on the unit cell dimensions; however, for comparison purposes, the number of cells was chosen as 4 x 4 x 1 cells in the x, y, and z directions, respectively, as shown in Figure 3. Nominal dimensions and weights were verified for each specimen before carrying out compression testing. For the compression tests, the cross-head displacement rate was set at 1 mm/min, loading cell recording the applied force.

Fig 2 - a) Tensile test setup for the NFC, b) flax/Psppawn composite geometry.
subjecting the three lattice structures to a uniaxial load in the z direction normal to the plane of the face sheet. Since the top and bottom of the specimen are flat, the loading plates transmit the force with no geometrical modifications. Whereas strut brackets like on the Lattice 1 and 3 specimen edges have been sanded down to create a flat surface on both the bottom and top ends on which the loading plates rest as illustrated in Figure 4.

Fig 3 - 3D printed PLA cellular lattice fabricated for compression test: a) Lattice 1, b) Lattice 2, c) Lattice 3, d) out-of-plane compression test setup.

Fig 4 - In-plane compression test setup: a) inclined struts sanded down, b) perfect alignment, c) load line direction with specimen under testing.
2.4.3 Shear testing of core lattices

A customised shear test set up fixture was designed to replicate the procedure described in the ASTM C273-00. Shear tests for the core structures were to determine their strength and modulus parallel to the plane of sandwich. The tests were conducted solely on the core material bonded to the thick loading plates. This method provides the load-deflection behaviour of core structures when loaded parallel to the facings. However, this method of testing does not produce pure shear as the secondary forces acting normal to the faces are present, but the prescribed specimen geometry and fixture setup reduce their effects on the obtained results. Test specimen dimensions were decided according to the ASTM recommendation as the three unit cell height were within the 10 mm envelope, 120 mm long core met the 12 times thickness minimum requirement and 60 mm width. The core adhesion to the steel plate was achieved using Araldite epoxy on the plate in the 50:50 by weight mixture and cured at 50°C for 4 hours to obtain the optimum lap shear strength.

The external load applied on the shear test rig produces a rotation of the steel plates that causes secondary peel stress at the interface with the core. Since the desired failure mode is 100% shear, specimens experiencing cohesive failures were rejected. Tensile load was applied on the rigid plate through the bolted bracket to the steel plates at crosshead displacement rate of 1 mm/min. Figure 5 illustrates the details of the testing methods.

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The contact areas of the core lattice structures have an effect on the quality of adhesion to the steel plates. In order to mitigate the premature debonding of the 3-D printed cores from the steel plates, the region adjacent to the bonding surfaces struts were modified. The diameter of the struts was increased only at the ends of the struts to double the contact areas. This modification was not needed for the compressive and flexural tests. The data for load-deformation can be used to derive the effective shear modulus and the ultimate shear strength using:

\[ \tau = \frac{P}{Lb} \] (1)
Where \( \tau \), \( P \), \( L \) and \( b \) are the core shear stress (MPa), load on specimen (N), length of specimen and width of specimen (mm), respectively. Equation 2 uses the specimen geometry and load-deflection response to derive the shear modulus \( G \) of the core:

\[
G = \frac{St}{Lb}
\]  

(2)

Where \( S \) and \( t \) are the slope of the initial portion of load-deflection curve (N/mm) and thickness of the core (mm).
2.4.4 Flexural testing of sandwich panels

Flexural behavior of the sandwich structures is affected by the ability of the core to withstand limits as it undergoes deformation as part of the overall beam displacement. Whereas the facings are responsible for the tensile and compressive load carrying providing direct stiffness and strength while being protected against local buckling. Three-point bending tests were conducted on the NF/3DP core sandwich panels according to ASTM C393-06 to determine their flexural properties. A TiNius Olsen testing machine equipped with a 25 kN load cell was used to perform the bending tests with a loading rate of 1 mm/min.

Sample preparation using the 3-D printed lattice cores permanently bonded to the NF facing sheets with Araldite 2015 epoxy resin in a 50:50 by weight mixture. Specimens with dimensions of 160 mm length, 40 mm width and overall thickness of 12 mm were put under testing with a span length of 125 mm, support and load nose radius of 5 mm as illustrated in Figure 6. The NF facing sheets were cut off a panel using a laser cutter in the above mentioned dimensions and 2 mm thickness. To avoid indentation failure within the facing sheets a facing/core ratio of 1:5 was adopted [43]. Sandwich beams were then placed in the oven for 4 hours for curing. The load $F$ and the crosshead displacement were recorded. Force-deflection curves were plotted, and subsequently flexural stiffness and failure load were determined. Three pieces of specimen per test condition were tested at room temperature 23 ± 2°C and 50% humidity.
The loading mechanism in terms of nose location with respect to the internal geometry of the core structure had an effect on the obtained results. In particular, a different effect was observed when the moving nose was placed directly on a vertical strut or when it was placed between two inclined struts, which was only a concern within Lattice 1 and 3. The effect of this positioning was quantified in the analysis.

## 3 Results and discussion

### 3.1 Tensile results of NFC

Figure 7a) shows the tensile stress vs strain curves of the flax reinforced PP. It is clear that the behavior of such material is heavily nonlinear as the tangent modulus reduces with increased strain. Three specimens were tested under the same conditions and produced repeatable results in terms of stiffness and strength. The failure of samples was initiated by matrix cracking due to increased strain, which consequently promoted the gradual decrease.
in stiffness at 0.5% strain. Also a closer look at the failed highlighted area in Figure 7b, it can be seen the partial failure of flax fibres under tension, which explains the change in stiffness after 0.5% strain. Also a closer look at the failed highlighted area in Figure 7b, it can be seen the partial failure of flax fibres under tension, which explains the extended load curve to strain exceeding 2%. The calculated Young's modulus on the initial part of the slope revealed behaviour of an average stiffness value of 1.6 GPa. It has been observed that the strength properties were affected by the fibre content of 52% as the fibre matrix interface dictates whether the fibre will improve the properties of the composites by transferring the applied load. Fibre distribution and geometry in terms of slenderness ratio also cause local variation of failure mechanisms. The strength of samples was considered as the highest point on the stress-strain curve as an averaged value of 15.2 MPa. The discontinuity and length variation of the flax fibres impede the proper distribution and alignment of load resulting in a shear-like effect hence lower properties. However, considering the strain energy demonstrated under the extended curves, few fibres holding the high load for extended strains as observed in Figure 7a within the upheld attachment across the failure area.

Figure 7a
Compressive results of core lattice structures

The load-displacement curves obtained from testing the three proposed lattice designs under flatwise and edgewise quasi-static compressions are summarized in Figures 8 and 9. It appears that the internal geometry of the lattice structures affected their behavior. The responses in all cases were elastic followed by either a plateau as load was maintained for higher displacement as observed within Lattice 1 or a dip of load after reaching the maximum over a longer displacement as observed in Lattices 2 and 3. Lattice 2 represented in Figure 8b depicted a variability over the three repeated measurements, which can be explained by the variability of specimen manufacturing as L2-S1 failure mechanisms of its vertical struts as global buckling is clearly observed in Figure 8d. A similar trend was observed in Lattice 3, as depicted in Figure 9.

Figure 8 - Load vs displacement curves obtained from the out-of-plane compression tests showing the three sample repetition denoted as S1, S2 and S3: a) Lattice 1, b) Lattice 2, c) Lattice 3 and d) L2_S1 deformation mechanism with vertical struts fracture.
The stiffness of the three structures was obtained from the slope of the load-displacement curve. It was observed that compressive stiffness of Lattice 1 out-of-plane was 15.5 kN/m whereas the highest stiffness was from Lattice 2 peaked at 18.02 kN/m and the lowest recorded was from Lattice 3 at 8.5 kN/m. The maximum peak load and stiffness values observed for similar lattice structures reported in literature [44, 45] are comparable to those measured for Lattice 1 and Lattice 2. However, the mechanical behavior of Lattice 3 is lower than the other two structures, although it is comparable to results reported in literature [46] for metallic cores. An effect of the internal geometry on the mechanical response of the overall structure can be clearly noticed in a way that the vertical struts tend to withstand high compressive loads at higher deflection. However, in some cases where the printing process has an effect on the performance, the vertical struts tend to buckle under high compressive loads, which explains the relatively maximum load carried by the structure before plummeting with no sign of the densification phase. The buckling mechanism as illustrated in Figure 8d initiates at mid-distance through the struts height which the trigo gse the overall structure side w y c lha pe. Co side ing L1 L2 a di L3 c nve s a ssho w in Figure 9, the re pe fo ma n e is te ms of stiffness s a di stre y h va de sc a he c d sifie din a de c eding o de ro fl2, L1 the n L3 a sthe c mc a l c ottributio nof the ve tic hstruts to w ds the struc utra linte gy wa s demo ntra e d
The in-plane compression tests performed on the L1, L2 and L3 revealed their structural differences in terms of stiffness and strength in addition to their failure mechanisms. All specimens show typical characteristics of cellular structures exhibiting an elastic response region, yield, plastic strain hardening up to a peak load and post-yield buckling or fracture. Overall, the maximum load taken by each structure was much lower compared to the out-of-plane in the range of 500 – 700 N. The in-plane testing direction is normal to the printing direction (i.e., layer by layer, therefore, the interlayer bonding strength is examined). L1 and L2 demonstrated higher strain energy capability as the load was maintained for higher displacement and an apparent ductile failure mechanism through the structure was observed across the three repetitions. In contrast, for which failure across all repetitions was consistently brittle with minimal deformation along the displacement direction. This can
be each by the missing vic struts, which helps the DL1 a DL2 with a ditio n of the missing vert struts, which helps the constitutive tense struts and the individual unit cell. Lattice samples do not have the connecting struts for load distribution flowing through the constituent struts of individual cells. Lattice samples tend to collapse layer by layer as the connecting struts fold triggering side movements of layers. It can be noticed that the slope and maximum load recorded for the tested specimen showed trivial variability with L1_S1 demonstrating the highest stiffness and strength. This can be related to the printing process. Structural imperfections within additional manufactured parts can be categorized into micro pores or non-uniform diameter of struts which directly affect the deformation mechanism within the lattice structure. Due to the less dense structure of Lattice 1 compared to Lattice 2, an elongated region of diagonal struts buckling before densification of the bottom row of cells was observed as illustrated in Figure 10a. Apart from Lattice 3 in which minimal deformation due to the post-yield hardening, which occurred at low strains followed by a sharp drop of load signifying a brittle fracture, Lattice 1 and 2 displayed strain-dominant deformation of the bottom layer. Lattice 1 shows layer-by-layer collapse through node rotation, while Lattice 2 shows shear band formation with stretch dominant deformation of the bottom layer.

Figure 10 - Deformation mechanisms of lattice structures under in-plane compression tests: a) Lattice 1 shows strain-plastructuring, and b) Lattice 2 displays strain-dominant deformation of the bottom layer.
3.3 Shear results of core lattice structure

The shear tests provide the equivalent stiffness and strength of the different core structures with their subsequent failure modes. The increased contact surface between the 3D printed core and the plate improves the adhesion strength and reduces the debonding effects induced due to the minimal bending effects. Common failure locations for all three lattice structures within strut joints as intersection points represent the weakest link of the structure as illustrated in Figure 11. Consistent results for all tests in terms of stiffness through load vs displacement repeatability, responses in all three configurations were linear elastic followed by either a brittle failure (Lattice 2 and 3) or gradual load drop at increasing strains. The shear strength and modulus have been computed from the experimental results using equation (1) and (2) and the apparent cross section area. It can be seen that Lattice 2 exhibited the highest shear capacities with maximum taken load of 16 kN equivalent to a shear strength of 2.3 MPa and shear modulus of 17.8 MPa. The sharp load drops within Lattice 2 and 3 after 1.5 mm displacement as the brittle failure of the overall structure at its critical connection points compared to the elongated displacement for Lattice 1 up to 3 mm with gradual loss of strength. Comparing the obtained results which are summarized in Figure 11d, it can be clearly noticed that the slope and maximum load (i.e., stiffness and strength) are affected by the internal topology of each lattice structure. The vertical struts within Lattice 1 provide structural stability under shear loading as the shear modulus is maintained at 17.8 MPa and a slight reduction of strength to 2 MPa as progressive failure of inclined struts under excessive bending causes the immediate load drop.
with a tensile failure of the connecting struts between adjacent unit cells. Figure 11b explains the overall response of Lattice 2 as the high load capacity is owed to the vertical struts taking most of it, but as the connecting struts within individual cells fail under a combined tensile and compressive modes, a subsequent abrupt load drop is associated with a horizontal dia glide close to the moving plate parallel to the loading plane. Although the similar features between Lattice 1 and 3 apart from the missing vertical strut, it is evident that Lattice 3 suffered a reduced shear strength and modulus of 1.7 MPa and 15.5 MPa respectively as the individual cells shifted along the displacement direction through the location of the weakest link at the intersecting node (Figure 11c). In addition to the latter, an evident rotation of diagonal struts as rupture occurs at both upper and lower nodes due to excessive bending.

Figure 11 - Load vs displacement curves from the shear test of bonded core lattice: a) Lattice 1, b) Lattice 2, c) Lattice 3 and d) Comparison of the three structures.
Failure mechanisms of the three core lattice structures under shear: (a) Lattice 1, (b) Lattice 2, (c) Lattice 3.
3.4 Flexural results of sandwich panels

Having investigated the compression and shear behaviour of the 3D printed core structures, the obtained results provide insight into the significance of bending loads on the sandwich structure. In flexural testing, the applied bending moment and transverse shear load are distributed between the components of a sandwich panel (i.e., face sheets and core, respectively). Figure 13 illustrates the load vs. deflection under the flexural test; three specimens have been tested under the same conditions to ensure the findings integrity. However, tests revealed inconsistency as testing of one lattice structures showed diverging responses in terms of flexural strength and plastic strains after the post-yielding point.

![Figure 13 - Bending characteristics of sandwich panels shown on the load vs. displacement. Three sample repetitions to show repeatability: a) Lattice 1, b) Lattice 2 and c) Lattice 3](image)

Figure 13 shows the load vs. displacement for the three lattice structures: a) Lattice 1, b) Lattice 2, and c) Lattice 3.

The overall response of all cell topologies was consistently highly nonlinear as the initial portion of the curve represented the flexural rigidity of the sandwich panel up to a point.
displacement of ~2 mm. The sandwich panels continue to withstand increasing loads under high plastic deformations beyond the elastic limit point up to a crosshead displacement of ~10 mm. It has been observed that the failure mechanism and flexural strength have been affected by the moving nose placement with respect to the internal geometry of the lattice structure.

In a setup in which the moving load is placed directly on a unit cell, an increased load-bearing capacity was observed with minimal deflection and consequent tensile failure of the bottom face-sheet. Whereas if the load is applied over the intersecting node of two adjacent cells, compression of the top face-sheet initiated folding into a V-like shape promoting the panel energy absorption at the cost of its flexural strength. Despite the latter, the stiffness have not been affected in which repeatability was validated by the consistent initial slope.

The averaged stiffness values recorded for L1, L2 and L3 were 142, 242 and 162 kN/m, respectively. Good skin-core adhesion was observed throughout all tested specimens as no evidence of delamination was observed, which implies that it can be eliminated as a variable to understand the performance.

Figure 14a shows Lattice 1 (L1_S1) local displacement of the diagonal strut along the loading line due to stretching whereas the bottom face-sheet exhibited tensile fracture. The maximum load of this particular test was 425 N, which was considered lower than L1_S2 (Figure 13a) exhibiting 462 N as the top face-sheet wrinkles at increasing deflection and maintained load. L1_S3 experienced similar response to L1_S1 at a maximum load held within an equivalent deflection region ~6.3 mm, however beyond 8 mm deflection load...

dro pe dthrou y htwo sta g e Tthis dissimilarity can be explained by high strength of the bottom face-sheet as the tensile fracture was less severe within L1-S3, hence the strut displacement distance. The progressive diagonal strut failure along the panel width on the normal plane to Figure 14 is associated with the near plateau load before the sharp drop of load. Results obtained from Lattice 2 tests were consistent in terms of the high stiffness and maximum load as well as the failure mechanism due to its dense topology. This has also reduced the wrinkling effect of the top face-sheet and promoted the tensile fracture of the bottom face and strut displacement at significantly higher loads ~765 N due to the relatively high density structure with vertical struts. Geometrical differences between L1 and L3 being the missing central vertical strut and increased diameter of all struts provided the latter with slightly improved stiffness and strength. The results reported in our paper are in coherence with several studies on the failure mechanisms of the sandwich structures components [47-49]. Although the reported studies focused on metallic truss cores, the parameters affecting the experimental programme are the same. Moreover, Zok et al. [50] developed mechanism maps for possible failures of pyramidal metallic truss cores based on beam theory with respect to face and core dimensions.
In the past, many researchers have studied the potential use of AM lattice structures for various applications [51-54]. Yoon et al. [55] investigated the compressive strength and modulus of 3-D printed ABS engineered trabecular bone and honeycomb biomimetic structures. Results proved that 3DP technology could be an efficient technique for comparative studies. Lee and Kang [56] carried out an experimental analysis on the compressive behavior of woven Kagome stainless steel wires as an aperiodic cellular metal structure within two steel face sheets; they found that as the number of layers increases along the out-of-plane direction, the strength decreases. In order to study the effects of variation in strut diameter on the compressive elastic modulus and collapse stress of cellular structures, Ravari et al. [57] fabricated PLA lattice structures using FFF for the mechanical assessment.

Fig 14: Deformation and failure mechanisms under bending loading: a) Lattice 1, b) Lattice 2, c) Lattice 3.

Figures not descriptions...
characterization and also developed beam and solid elements based FEM. To achieve realistic elastic modulus and strength each strut needed to be divided into ten and twelve equally spaced intervals respectively. The results found in this study complement these past works as they can be used to predict mechanical properties of appropriately modelled sandwich structures.

The maximum load values obtained from averaging the three maximum load recorded from samples repetition have been normalised with the nominal weight and reported in Table 2. From the comparison, the diamond structure with vertical strut (Lattice 1) performs the best in compression out of the plane and shear whilst the diamond structure without vertical strut (Lattice 3) performs better under compression in plane and three-point bending load. However, the latter diamond structure was modified increasing the diameter of the struts from 1.5 mm to 2 mm. This change justifies the increase in normalised maximum load for in-plane and flexural since for the diagonal struts are primarily contributing to the bending and in-plane strength of the cell.

Table 2 - Maximum load values normalised with the nominal weight (N/g).

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<tr>
<th></th>
<th>Normalised maximum load* F/m(N/g)</th>
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<tbody>
<tr>
<td></td>
<td>Lattice 1</td>
</tr>
<tr>
<td><strong>Compressive out-of-plane</strong></td>
<td>5 0 9 0</td>
</tr>
<tr>
<td><strong>Compressive in-plane</strong></td>
<td>2 9 0</td>
</tr>
<tr>
<td><strong>Shear</strong></td>
<td>8 3 0</td>
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<tr>
<td><strong>Flexural</strong></td>
<td>3 2</td>
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</table>

These results are promising. However, there are several limitations to this study that can be improved in future work. First, the top and bottom sheets made of PP-flax are limited to a to this study that the mechanical properties of the P-Pfla x
Conclusions

In this paper, FFF technology was used to manufacture lattice structures used as cores for biocompatible sandwich panels with natural fibre composite skins. The mechanical properties of 3D printed PLA cores under compression both in-plane and out-of-plane, shear and flexural of the biocomposite panels have been carried out. The key findings established in this work are listed as follows:

1. The skin itself is a PA laminate using a high-resolution nozzle that is inadequate for timely manufacturing. Future work should investigate large layer thicknesses from a large nozzle to enable rapid manufacturing of biocomposite sandwich structures.

2. Work should expand applications beyond flat biocompatible sandwich panels with natural fibre composite skins to more advanced shapes where the skin itself could be 3D printed.
A comparison between the investigated geometries shows that the diamond structure with vertical strut (Lattice 1) performs best in compression out of the plane whilst the diamond structure without vertical strut (Lattice 3) performs better under compression in plane and three-point bending load. The better response for the diamond structure was achieved through modifying the strut diameter from 1.5 mm to 2 mm since the diagonal struts are primarily contributing to the bending and in-plane strength of the cell.

The tensile tests on the natural fibre showed that the failure was ductile with matrix failure first followed by stretching of the fibres until complete failure. The failure mechanisms of the core structures observed under the different loading conditions have been discussed. Under compression, in-plane shear band and rotation of the struts have been observed. For the compression out of plane, the main mechanism is plastic hinges followed by densification. In some cases buckling of the vertical struts has been noted. Under shear load, the failure generally occurred at the nodes where the diagonal struts intersect the vertical struts. In addition for the diamond structure without vertical strut, the failure also starts in the middle plane at the intersecting nodes. Under flexural load, the failure always appeared in one of the lower diagonal struts closer to the middle plane under tension load.

On the basis of the experimental results, the three shapes investigated are well suited for impact applications because of the high shear and out of plane compression strength.
The experimental results discussed in this work prove the feasibility of AM technology in the process of manufacturing lightweight polymer-based sandwich panels for structural applications. Based upon experimental results this category of core structures could compete with high performing honeycomb structures used for aerospace applications. However, as quality assurance of parts produced with this technology, repeatability is still an issue to be addressed.

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