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Graphene Photonic Crystal Fibre with Strong and Tunable Light-Matter Interaction

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26 **The integration of photonic crystal fibre (PCF) with various functional materials has greatly**
27 **expanded the application regimes of optical fibre¹⁻¹². The emergence of graphene excites new**
28 **opportunities by combining with PCF, allowing for electrical tunability, broadband optical**
29 **response and all-fibre integration ability¹³⁻¹⁸. However, the previous demonstrations are typically**
30 **limited to the sample level of micron size, far behind the requirement of real applications for the**
31 **metre-scale material level. Here, we demonstrate a new hybrid material of graphene photonic**
32 **crystal fibre (Gr-PCF) with length up to half a metre by chemical vapour deposition method. The**
33 **Gr-PCF shows strong light-matter interaction with ~ 8 dB \cdot cm⁻¹ attenuation. In addition, the**
34 **Gr-PCF-based electro-optic modulator demonstrates broadband response (1150 - 1600 nm) and**
35 **large modulation depth (~ 20 dB \cdot cm⁻¹ at 1550 nm) under low gate voltage of ~ 2 volts. Our results**
36 **could enable industrial-level graphene applications based on the Gr-PCF, and suggest an infusive**
37 **platform of two-dimensional material-PCF.**

38 Graphene is a promising material in photonic and optoelectronic applications due to its superior
39 properties of high carrier mobility, broadband optical response and facile electrical tunability originating
40 from its unique linear dispersion of massless Dirac fermions¹⁹⁻²⁸. Although the light-matter interaction in
41 graphene normalized by its atomic thickness (0.34 nm) is quite strong, the measurable interaction is in
42 fact quite weak (only $\sim 2.3\%$ light absorption)²⁹. To greatly enhance light-graphene interaction, many
43 efforts have been devoted to combine graphene flakes with well-designed optical structures, such as
44 gratings, waveguides and microcavities³⁰⁻³⁴, however, all those hybrid structures have still stayed at
45 sample level of micron size, rather than material level of metre size, which limits their massive
46 applications. Therefore, there exists great demand to develop new methods for massive production on
47 graphene-based optical structures for material-level applications.

48 Optical fibre provides the highest-quality optical waveguide for information communication and
49 photon manipulation, and it has been massively manufactured at kilometre length scale. PCF represents
50 the most important advance of optical fibre in the last twenty years and possesses extremely rich
51 functions beyond traditional optical fibre in the exciting applications of endlessly single-mode fibres,
52 supercontinuum lasers, frequency combs, optical soliton propagation, high-power pulse delivery and so
53 on¹⁻⁷. Especially, PCF with ingenious porous structure opens up the hard-won opportunity of filling
54 various materials, ranging from gases, liquids, solids to liquid crystals, to expand its great new

55 functionalities in mode-locked fibre lasers, laser frequency conversion, surface plasmon generation,
56 stimulated Raman scattering and in-fibre thermal- or electro-optic devices⁸⁻¹⁵. The rise of
57 two-dimensional (2D) graphene naturally excites the keen interests in combining PCF with graphene.
58 Such graphene-PCF complex can occupy several unique advantages: *(i)* the flexibility of graphene
59 facilitates its tight attachment to the hole walls of PCF; *(ii)* the atomic thickness of graphene keeps the
60 PCF structure and main optical function intact, and *(iii)* the distinct properties of graphene bring the
61 unique functions that can't be realized by any other conventional materials. Indeed, great efforts have
62 been delivered to fabricate graphene-optical fibre complex by transferring graphene flakes on
63 side-polished or tapered normal optical fibres¹⁶⁻¹⁸, or filling them into the holes of PCFs¹³⁻¹⁵. However,
64 all these attempts are only at sample level of micron length with small interaction area and harmful
65 treating on fibre modes, and it is still lacking an efficient and nonharmful manufacture strategy for the
66 massive production at metre-sized material level for graphene-PCF complex with large interaction area
67 and intact structure yet.

68 In this work, we report the production of the new hybrid material of Gr-PCF with half-metre length
69 by direct chemical vapour deposition (CVD) growth method, which was previously believed extremely
70 challenging due to the lack of metal catalyst and the difficulties in gas flow control along the long
71 micron-sized holes in silica PCF. Our success herein benefits from our extensive experience in the
72 growth of graphene glass³⁵ and the well control of molecular gas flow in the confined space³⁶. With this
73 new Gr-PCF material, we realized greatly enhanced light-matter interaction between graphene and
74 core-guided light with $\sim 8 \text{ dB}\cdot\text{cm}^{-1}$ transmission attenuation. We also demonstrated the tunable
75 light-matter interaction by electrically gating graphene through ionic liquid, with large modulation depth
76 ($\sim 20 \text{ dB}\cdot\text{cm}^{-1}$ at 1550 nm) and broadband wavelength response (1150 - 1600 nm) under low gate voltage
77 (~ 2 volts). And the fully enhanced and tunable light-matter interaction in Gr-PCF material indicates its
78 great potentials in the all-fibre integration devices. Our results pave a new way for hybrid optical fibre
79 manufacturing and suggest an exciting material platform of 2D materials-integrated fibre with
80 unprecedented function tunability in both linear and nonlinear optics.

81 In our experiment, Gr-PCF was grown with methane (CH_4) as carbon feedstock flowing through the
82 PCF's narrow holes ($\sim 4 \mu\text{m}$ in diameter) under $\sim 1100 \text{ }^\circ\text{C}$ with controlled pressure (Fig. 1a). After the
83 growth, the PCF can keep its structure intact which consists of a solid silica core and a surrounding

84 cladding region with a patterned array of holes (Fig. 1b), while graphene is grown on both the outer
85 surface and inner hole walls of PCF. The fully-covered graphene on the outer surface was
86 unambiguously evidenced by the Raman mapping (Fig. 1c and Supplementary Fig. 1). Representative
87 Raman spectrum (Fig. 1d) shows sharp G- and 2D-mode peaks, revealing the high quality of graphene
88 (The observable D peak mainly comes from the defective grain boundaries between graphene
89 domains)³⁷. The fully-covered graphene on the inner hole walls can be firstly indicated by the darker
90 optical contrast of Gr-PCF than bare PCF, due to the light absorption by graphene (Fig. 1e,f). Direct
91 evidence of graphene growth inside PCF can be given by the observation of tube-like graphene
92 frameworks that tightly attach on the hole walls and protrude out the holes when one breaks the Gr-PCF
93 (Fig. 1g and Supplementary Fig. 2). Furthermore, the inner graphene films can be directly obtained by
94 dissolving the fibre silica in hydrofluoric acid (the graphene film on the outer surface was first removed
95 by air-plasma treatment). The cylindroid graphene films from Gr-PCF collapse into ribbon films on
96 silicon substrate, with an average layer thickness of ~ 2.0 nm as measured by atomic force microscope
97 (AFM) (Fig. 1h,i). Here we note that the interlayer distance is relatively larger (estimated as 1.0 - 1.5 nm)
98 than its intrinsic interlayer distance (0.34 nm) in these collapsed ribbons, likely due to the wrinkles
99 formed during the etching treatment. The selected-area electron diffraction (SAED) in Fig. 1j
100 demonstrates the polycrystalline structure of the graphene film, as the selected-area aperture size (200
101 nm) is several times of graphene domain size (~ 50 nm). However, the high-resolution transmission
102 electron microscopy (HRTEM) image on individual graphene domain shows perfect graphene lattice
103 structure (Moiré pattern of bilayer as an example in Fig. 1k). More transmission electron microscopy
104 (TEM) characterizations reveal that the graphene thickness can be controlled between one to ten layers
105 by the growth time (Supplementary Fig. 3).

106 One great concern about the Gr-PCF growth is whether the graphene can be homogeneously
107 distributed on the micron-sized hole walls along the long fibre, as gas flow in narrow space will have
108 huge viscous force. To clarify this point, we carried out control experiments with atmosphere pressure
109 chemical vapour deposition (APCVD, Fig. 2a) and low-pressure chemical vapour deposition (LPCVD,
110 pressure of 0.5 - 1.0 kPa, Fig. 2b). According to viscous flow model³⁸ (Supplementary Note 1), the mean
111 free path of carbon precursors (estimated as ~ 0.4 μm) is much shorter than the fibre hole diameter (~ 4
112 μm) under the atmosphere pressure in APCVD process. In this situation, the carbon precursor feeding is

113 limited by the mass transfer³⁶. According to experimental Raman data of graphene ribbons etched from
114 the holes of Gr-PCF with ~8 cm length, the obvious variation of 2D- to G-mode Raman (intensity ratio
115 I_{2D}/I_G , 0.6 - 2.1) and full width at half maximum of 2D-mode ($FWHM_{2D}$, 31 - 71 cm^{-1}) demonstrate the
116 nonuniform thickness along the fibre axis (Fig. 2c,d, APCVD panels). This result is consistent with
117 optical contrast observation that the fibre became darker along the gas flow direction (Supplementary
118 Fig. 4, lower fibre). It means the graphene film thickness increases along the gas flow, originating from
119 the increased active carbon species during the long-time thermal decomposition in the downstream
120 positions. Whereas, in LPCVD process, the gas flow approaches a free molecular flow situation and the
121 mass-diffusion process becomes negligible^{36,38}. In this case, the mean free path (estimated as $> 40 \mu m$) is
122 much larger than hole size of PCF. Then elapse time (~15 s) of gaseous molecules going through the
123 long holes of PCF in LPCVD process is much shorter than that in APCVD process (~175 s). It is facile
124 to realize that the graphene films on the inner hole walls grow with uniform thickness along the fibre
125 axis, as revealed by the Raman spectra with negligible fluctuation of I_{2D}/I_G ratio (~1.4) and $FWHM_{2D}$
126 (~48 cm^{-1}) (Fig. 2c,d, LPCVD panels), as well as the uniform optical imaging contrast along the whole
127 fibre (Supplementary Fig. 4, upper fibre). Here we note that the intensity ratio of D- to G-mode Raman
128 (I_D/I_G) has much larger value in LPCVD, revealing smaller graphene domain size³⁷, which originates
129 from the lower carbon feedstocks, corresponding to the slower growth rate and the more nucleation
130 centres under molecular flow. Using current furnace with heating zone of ~60 cm, we can readily grow
131 Gr-PCF with uniform thickness and length up to 50 cm (Fig. 1e).

132 In principle, one would expect that the light-graphene interaction in Gr-PCF would be greatly
133 enhanced as the atomic thickness of graphene wouldn't destroy the fundamental propagating mode. This
134 assumption was checked by simulating the light electric field distribution of fundamental guiding mode
135 with the full-vector finite element method³⁰. In the simulation, the graphene film directly interacts with
136 light field by evanescent wave coupling at the hole walls adjacent to the fibre core (Fig. 3a). Similar to
137 the distribution in bare PCF (Supplementary Fig. 5d), the light in Gr-PCF is mainly confined in the fibre
138 core and about one tenth of the electric field to that at fibre core centre is interacting with graphene on
139 the innermost hole walls (Fig. 3b). The light-graphene interaction can be first indicated by the 3% kink
140 in the normalized radial electric field distribution, which originates from refractive index difference
141 between graphene/silica and air. The significantly enhanced light-graphene interaction in Gr-PCF can be

142 quantitatively measured by the propagated light intensity evolution along the fibre. In a bare PCF of ~ 4
143 cm long, there is no observable attenuation ($< 0.01 \text{ dB}\cdot\text{cm}^{-1}$) at all (Fig. 3c, purple dots). However, in
144 Gr-PCF, a strong attenuation of $8.3 \text{ dB}\cdot\text{cm}^{-1}$ is observed (Fig. 3c, cyan dots, fitted slope), in striking
145 contrast with only 0.1 dB of suspended monolayer graphene. The attenuation coefficient value of this
146 Gr-PCF is equivalent to average graphene layer number of ~ 1.5 (Fig. 3c, line and Supplementary Fig.
147 5i). Here the 0.5 layer comes from the partial second graphene layers grown on the first graphene layer
148 film. The greatly enhanced light-graphene interaction in Gr-PCF can be qualitatively understood by the
149 tremendous enlargement of effective interaction area and length during the multiple reflection of light
150 propagation along the fibre axis (Fig. 3c, lower panel). It is worth noting that the light-graphene
151 interaction strength can be tuned by the PCF geometries, such as the air-hole diameter and the hole pitch
152 size (Supplementary Fig. 6).

153 One of the unique merits of 2D graphene is the unprecedented electrical tunability in light-matter
154 interaction by shifting its Fermi level (E_F)^{21,26}. This tunability can be used to exploit the Gr-PCF as an
155 in-line electro-optic modulator with intensity modulation in all-fibre communication networks (testing
156 setup in Supplementary Fig. 7). Note that our Gr-PCF shows a stable and negligible loss ($< 0.1 \text{ dB}$) in
157 coupling with single mode fibres. Here, we demonstrate an electro-optic modulator consisting of a short
158 segment (0.5 - 1.5 cm) of Gr-PCF with ionic liquid (DEME-TFSI) filled inside the holes (Fig. 4a). The
159 graphene on the hole walls of Gr-PCF connects electrically with an electrode on the outside graphene
160 surface (the as-grown graphene films in the whole fibre are all connected). Ionic liquid is then fully
161 filled in the fibre holes and contacts with another electrode. When gate voltage between graphene and
162 ionic liquid is applied, electrical double layer (EDL) will form at graphene-ionic liquid interface and
163 dope graphene efficiently under low gate voltage of several volts (Fig. 4b and Supplementary Fig.
164 8)^{26,39,40}. Further simulation reveals an additional advantage of ionic liquid filling that it can increase the
165 light-graphene interaction from 5 to $24 \text{ dB}\cdot\text{cm}^{-1}$ for monolayer Gr-PCF by increasing the mode field area
166 of core-guided light (Supplementary Fig. 5), as ionic liquid has a refractive index (1.42) close to that of
167 the PCF material (silica, 1.44).

168 By electrically tuning graphene Fermi level via ionic liquid gating, the interband transition
169 absorption can be tuned at “On” or “Off” state when E_F is smaller or larger than half photon energy of
170 $\hbar\omega/2$, where ω and $\hbar\omega$ are the angular frequency and energy of a photon, respectively (Fig. 4c)^{21,30}. This

171 absorption-based electro-optic modulator operates at very low voltage (within 2 V) due to the efficient
172 electric double layer gating, and a broadband spectral response from 1150 to 1600 nm due to the linear
173 Dirac band structure of graphene (Fig. 4d). In particular, at the fibre-optic communication O- (1310 nm)
174 and C-wavelength bands (1550 nm), the modulation depths can respectively reach ~ 13 and ~ 20 dB \cdot cm $^{-1}$
175 at gate voltage of -1.8 V (Fig. 4e). The relatively lower gating efficiency at positive gate voltage region
176 is attributed to the lower capacitance of our ionic liquid at electron doping side, which is consistent with
177 the electrical device measurement (Supplementary Fig. 8). The performance of our current modulator
178 can be further improved. Firstly, it has insertion loss of ~ 6 dB/cm in the “On” state, which can be much
179 lower in principle if one grows very high-quality monolayer graphene on the inside walls of PCF.
180 Secondly, it has slow switching speed of ~ 16 Hz (Supplementary Fig. 9) which can be improved (e.g.,
181 few 10s GHz^{17,31}) if one can achieve optimized structures, such as Gr/hBN/Gr-PCF⁴¹. Nevertheless, the
182 Gr-PCF modulator has high potential to be integrated with other optical devices in all-fibre system
183 where light can be synchronically transmitted, modulated and detected inside fibre link without the aid
184 of discrete devices in the future⁴².

185 In summary, we demonstrated an ingenious CVD route to achieve a new material of Gr-PCF. The
186 uniform and high-quality graphene on both the outer surface and inner hole walls of PCF with length up
187 to half a metre was realized by low pressure growth with controlled molecular flow. Such Gr-PCF
188 material exhibits a strong and tunable light-matter interaction and presents excellent performance as a
189 broadband electro-optic modulation under low gate voltage. As added convenient electrical tunability,
190 kept PCF waveguide mode intact and enhanced light-graphene interactions in both linear and nonlinear
191 optical regime, Gr-PCF enables the potential advent of novel fibre-optic devices such as
192 electrically-tunable mode-locked all-fibre lasers, gate-controllable wavelength-independent nonlinear
193 wavelength converters, tunable broadband polarizers and optical limiters in the near future. In addition,
194 our growth strategy can open up a new direction for massive production of other 2D crystals based
195 all-fibre devices, targeting to the next-generation optical fibres with various new functionalities.

196 **Online content**

197 Any methods, additional references, Nature Research reporting summaries, source data, statements of
198 data availability and associated accession codes are available at <https://doi.org/xxx/xxx>

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291 **Author contributions**

292 Z.L. and K.L. supervised the project. Z.L. and K.C. conceived the material growth. K.L. and X.Z.
293 conceived the optical measurement. K.C. and X.Z. carried out the material growth experiment and
294 optical device measurements. X.C. performed theoretical modelling. K.C., R.Q., Y.C., Y.X., X.Z., C.L.
295 and F.Y. conducted SEM, TEM, AFM and Raman characterizations. W.Y. suggested the optical
296 experiments. F.Y. programmed the measurement software. Z.S. and F.W. suggested the modulator. All
297 authors contributed to the scientific discussion and writing of the manuscript.

298 **Competing interests**

299 The authors declare no competing financial interests.

300 **Additional information**

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304 optics).

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308 **Figure captions:**

309 **Fig. 1 | Growth and characterization of Gr-PCF.** **a**, Schematics of Gr-PCF grown by CVD method,
310 with graphene film on both outer surface and inner walls of PCF. **b**, SEM image of Gr-PCF end surface.
311 **c,d**, 2D-mode Raman intensity mapping (**c**) and Raman spectrum (**d**) of graphene at fibre end surface. **e,f**,
312 Optical reflection photograph (**e**) and transmission micrograph (**f**) of bare PCF and Gr-PCF. The optical
313 contrast is darker for Gr-PCF due to the light absorption by graphene. **g**, SEM image of a tube-like
314 graphene protruding out one hole of the fractured Gr-PCF. **h**, SEM image of a graphene ribbon (the
315 collapsed tube-like graphene grown on the hole wall) after dissolving the fibre silica. **i**, AFM image of a
316 graphene ribbon with wrinkles. The height of ~ 2.0 nm is obtained along the white dash line. **j,k**, SAED
317 (**j**) and HRTEM (**k**) images of the graphene ribbon.

318 **Fig. 2 | Controlled growth of uniform graphene film on the hole walls of PCF.** **a,b**, Schematics of
319 graphene growth on the walls of the micron-size holes of PCFs at atmosphere pressure (AP) and low
320 pressure (LP) CVD. **c**, Representative Raman spectra of graphene at different positions of graphene
321 ribbons along the gas flow under APCVD (upper panel) and LPCVD (lower panel) growth conditions.
322 The graphene ribbons are from the hole walls after etching away the fibre silica. **d**, The statistics of I_D/I_G ,
323 I_{2D}/I_G and the $FWHM_{2D}$ of graphene Raman peak at different positions of graphene ribbons by APCVD
324 and LPCVD, where D, G and 2D stands for D-, G- and 2D-mode Raman peaks of graphene, respectively.

325 **Fig. 3 | Strong light-matter interaction in Gr-PCF.** **a**, The simulated light electric field distribution of
326 the fundamental guiding mode at 1550 nm. The graphene is set on the hole wall shown in the profiles of
327 Gr-PCF (low panel, yellow rectangle). **b**, The distribution of normalized electric field in Gr-PCF along
328 horizontal radial direction (yellow horizontal line in (a)). The right panel shows the zoom-in of the dash
329 square in left panel. The black arrow highlights the graphene position. **c**, Measured optical attenuation of
330 light propagation in bare PCF (purple dots) and Gr-PCF (cyan dots) with different fibre lengths. The
331 attenuation coefficient is fitted as $8.3 \text{ dB}\cdot\text{cm}^{-1}$ of the slope. The lower panel gives the schematic of light
332 attenuation with multiple reflection during its propagating along Gr-PCF core. The error bar is the
333 standard deviation from the measurement of 5 samples for each length grown at the same growth
334 conditions.

335 **Fig. 4 | Tunable light-matter interaction in Gr-PCF.** **a**, Schematic of a Gr-PCF-based electro-optic
336 modulator. The gate voltage between ionic liquid and graphene controls the light transmission through
337 Gr-PCF. **b,c**, The working principle of Gr-PCF electro-optic modulator where electric double layer
338 forms at interface between graphene (dark grey rectangle) and ionic liquid (blue area). The ionic
339 liquid-gating tunes the graphene's Fermi level and switches on and off the optical absorption in graphene.
340 When $E_F < \hbar\omega/2$ ($E_F > \hbar\omega/2$), graphene absorbs (doesn't absorb) light and the modulator is working at
341 "Off" ("On") state for light transmission in the fibre. **d**, Two-dimensional mapping of transmission
342 modulation (normalized by fibre length) of Gr-PCF modulator as a function of gate voltage and optical
343 wavelength. **e**, The modulation curves at 1310 and 1550 nm show an unambiguous transition between
344 "On" and "Off" state with large modulation depth.

345 **Methods**

346 **Growth of Gr-PCFs.** The PCFs (125 μm of cladding diameter and 8 μm of core diameter) were placed
347 in the centre of a tube furnace (Thermo Linderberg). For APCVD process, CH_4 (10 sccm) was
348 introduced as a carbon feedstock with Ar (100 sccm) and H_2 (50 sccm) at 1100 $^\circ\text{C}$ for hours under
349 atmosphere pressure. For LPCVD process, CH_4 (50 sccm) was mixed with the H_2 (50 sccm) at 1100 $^\circ\text{C}$
350 for hours under pressure of 500 - 1000 Pa. After the growth, the as-grown sample was naturally cooled
351 down in the protective gases of Ar and H_2 .

352 **Characterization of Gr-PCFs.** An optical microscope (Olympus BX51) was used to obtain the optical
353 images of Gr-PCFs. SEM images were collected by FEI NovaTM Nano-SEM430 at 5 kV. Raman spectra
354 and mappings were taken by WITec alpha 300R-confocal Raman imaging system equipped with a 532
355 nm laser. AFM pictures and data were performed in a Bruker Dimension Icon atomic force microscope.
356 HRTEM and SAED experiments were performed in an aberration-corrected FEI Titan Themis G2-300 at
357 80 kV.

358 **Numerical modelling.** Numerical simulations were processed by the RF module of COMSOL's
359 Multiphysics software and MATLAB software. The effective refractive index of fundamental guiding
360 mode in Gr-PCFs was calculated by finite element method in COMSOL.

361 **Device fabrication and modulation measurement.** Ti/Au (5/100 nm) electrodes were deposited on
362 graphene film by e-beam evaporation. The ionic liquid (DEME-TFSI, Sigma-Aldrich, CAS:
363 464927-84-2) was injected into holes of Gr-PCF with another Ti/Au electrode connection. The two ends
364 of Gr-PCF were connected with two single-mode optical fibres (SMFs, Corning SMF-28e+) and the gap
365 between fibres were filled with ionic liquid. The two SMFs were aligned in line with Gr-PCF by our
366 home-built setup (Supplementary Fig. 6), where two optical microscopic observation in different
367 directions and multi-axis piezo-stages ensured the accurate alignment. Keithley 2400 Source Meter
368 provided the gate voltage. Supercontinuum laser (NKT Photonics Inc.) was coupled into SMF to provide
369 broadband light. Modulated signal was sent into a spectrograph equipped with an infrared CCD
370 (Princeton Instruments: Acton SpectraPro® SP-2300 and Pylon-IR 1024-1.7).

371 **Data availability**

372 The data that support the plots within this paper and other findings of this study are available from the
373 corresponding author upon reasonable request.

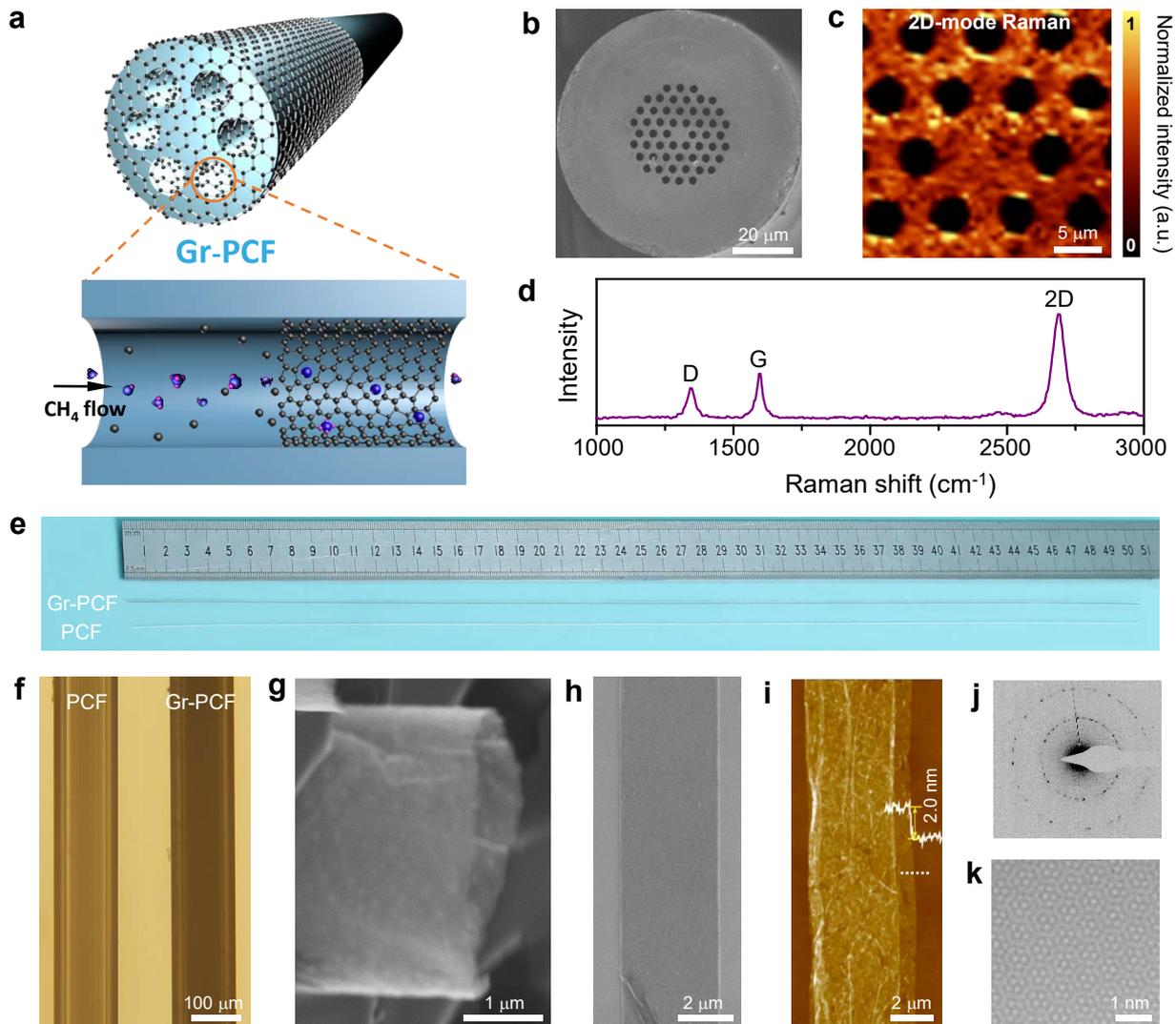


Fig.1

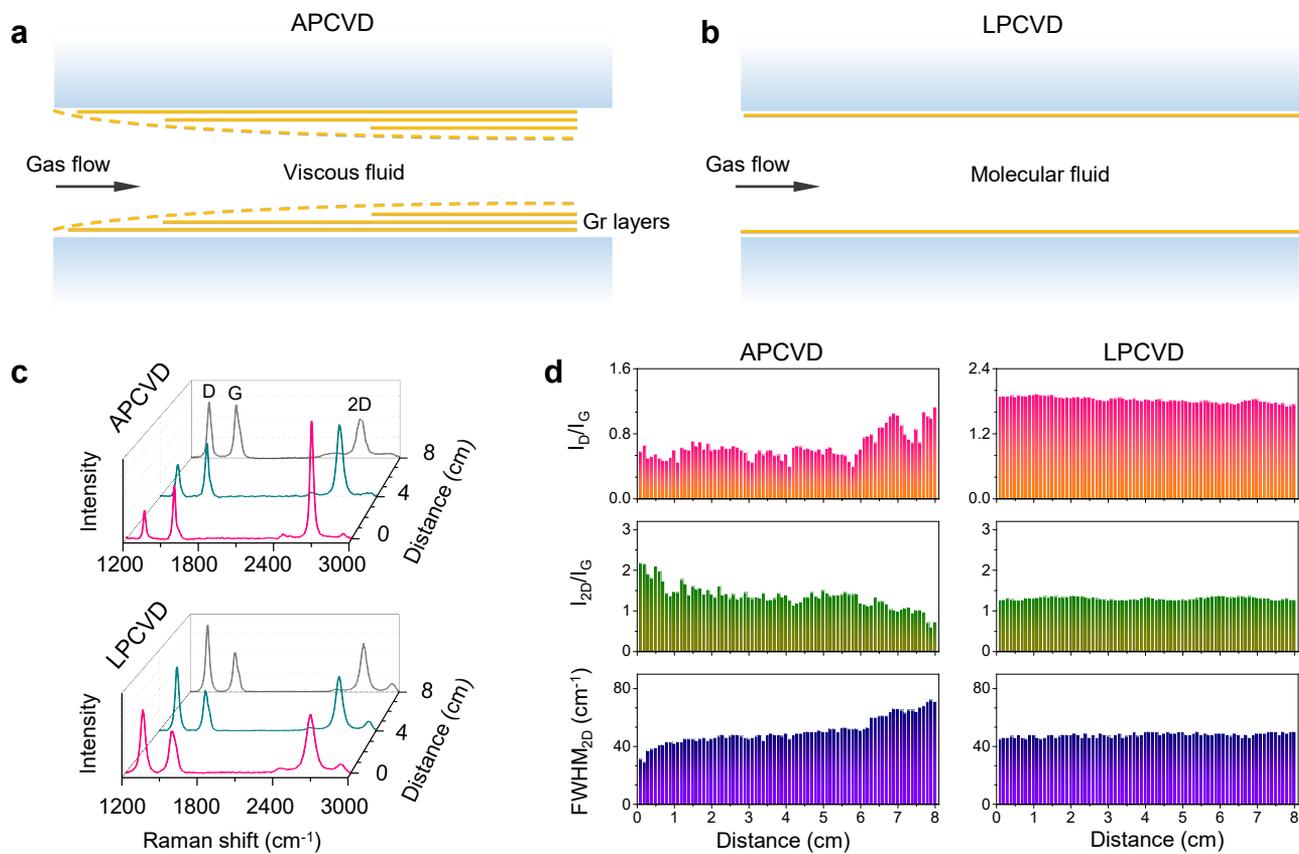


Fig.2

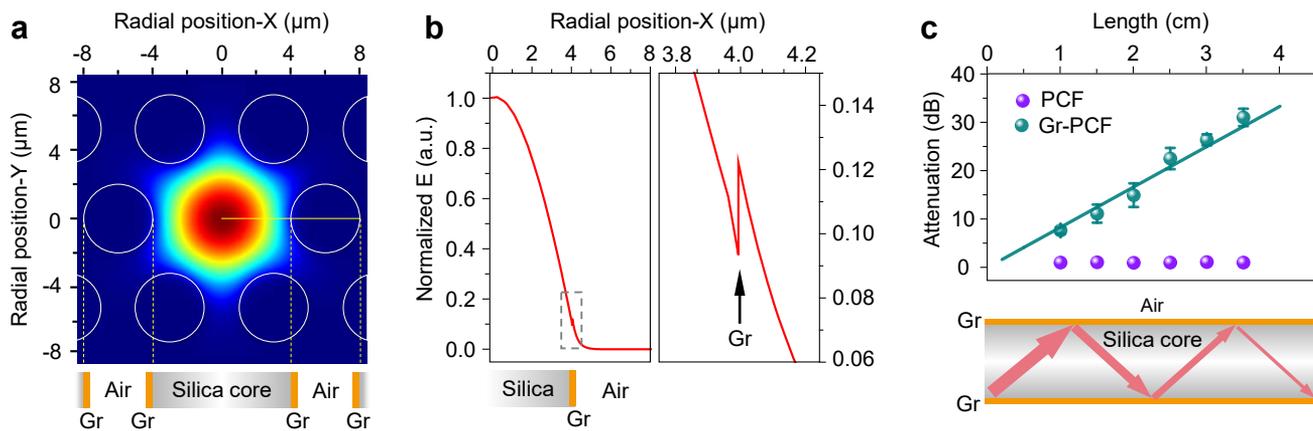


Fig.3

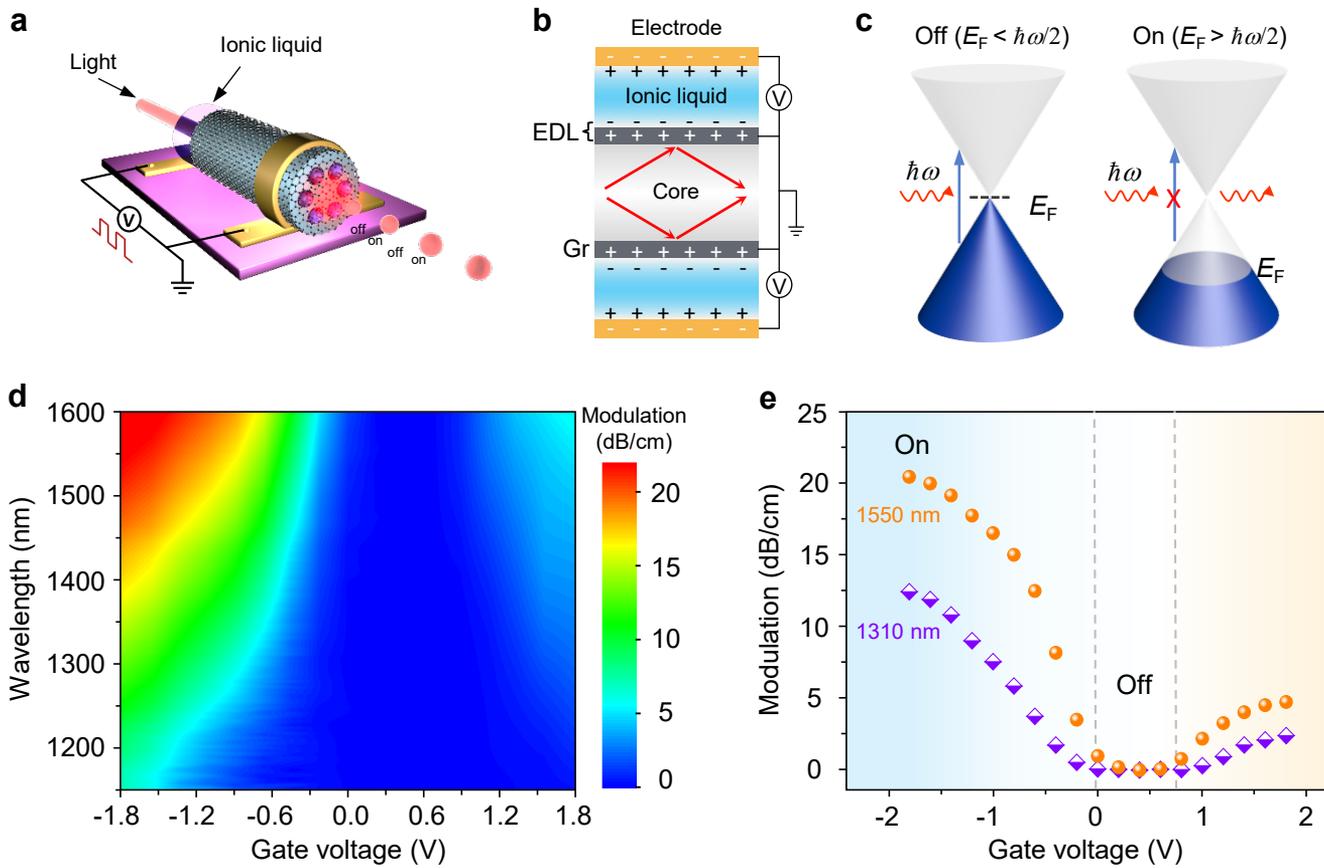


Fig.4