Panchal, Mrunal; Patel, Dev; Vyas, Hardik; Mehta, Kush

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Ultra-thin Friction stir welding on Aluminum alloy
Mrunal Panchal, Devax Patel, Hardik Vyas, Kush Mehta

1Department of Mechanical Engineering, School of Technology (SOT), Pandit Deendayal Petroleum University (PDPU), Raisan, Gandhinagar 382355, Gujarat, INDIA
2Advanced Manufacturing and Materials group, Department of Mechanical Engineering, School of Engineering, Aalto University, Finland

Abstract

Ultra-thin sheets’ welding is an intricate process either by conventional or advanced techniques. In the present investigation, Micro friction stir welding (FSW) was successfully performed to join the ultra-thin (0.5 mm thick) commercial aluminum. Suitable fixture and tool were developed for the ultra-thin sheets to perform experiments. Velocity ratio of rotational speed to transvers speed was varied to obtain ultra-thin friction stir welding. The welded samples were investigated by visual examination, macrographs, microstructure, tensile testing with fracture surface analysis and micro hardness distribution. The results revealed that, the sound joint was achieved at the velocity ratio of 13.71. The tensile strength of 100.877 N/mm² and elongation of 23.12% were obtained, which were 90.77% and 51.377% of the parent material respectively. Fracture surfaces after tensile testing was observed with elongated dimples indicating ductile fracture. The maximum micro hardness of 94 HV was observed in the weld zone.

Keywords: Aluminum alloy, Mechanical Properties, Micro friction stir welding, Microstructure, Ultra-thin

1. Introduction

Friction stir welding (FSW) was developed in 1991 by Wayne Thomas and his colleagues in The Welding Institute (TWI) [1, 2]. FSW is a solid-state welding process that works on visco plastic material mixing caused by thermo-mechanical processing [3]–[6]. FSW has least impact on the environment due to solid state green welding that does not partake in any fumes, red-hot glowing materials, spattering or a loud noise during operation [7]. In FSW, a non-consumable tool made up of suitable features and dimensions of shoulder and pin is plunged into the area to be joined. Due to plastic deformation and friction between surfaces of tool and workpiece produces heat [8]-[10]. The material mixing with fine and recrystallized grains is established by stirring that deformed material with a favorable combination between rotational and transverse velocities of non-consumable tool.

FSW is invented for Al and its alloys that is further expanded for high temperature materials, dissimilar materials, composites, and plastics [11]. FSW is under development with its novel variants for different size of workpieces for aforementioned materials. Micro FSW is one of such variant to obtain weld having thickness less than 1 mm. Micro FSW offers advantages such as I) It significantly increases the possibility of joining wider range of similar and dissimilar material (plasticsisable materials), II) Joints are free of any contamination as there is no use of filler materials and fluxes, III) Difficult to weld Aluminum alloys can be welded without using any shielding gas [1, 2]. Sithole and Rao [12] presented a review on recent developments in micro FSW with its importance and process details.

In 2004, Nishihara and Nagasaka [13] revolutionized FSW by introducing welding of thin-plate structural material of sub 1 mm materials and named that as micro FSW. Micro FSW is further investigated by Taylor et al. [14] in 2008 with successful welds of 0.8 mm Aluminum (Al) alloy. Baskoro et al. [15] studied the effect of welding parameters for micro FSW in order to better understand the process-parameters influence on weld formation for Aluminum alloy.

[copyright information to be updated in production process]
in lap configuration with 0.4 mm thickness. The same group of Baskoro et al. [16] in 2015 performed micro FSW with experimental studies to provide understanding on friction stir spot welding for parameters-process influence on joint formation. Wanshon and Binti [17] developed understanding on heat dissipation in case of micro FSW using different backing plates namely a ceramic anvil backing plate and a conventional stainless steel anvil backing plate. Papaefhyimo and Goulase [18] performed the feasibility test of micro FSW for titan zinc plates (ZnTiCu) having a thickness of 700 µm, wherein a low velocity ratio with tool rotation speed of 1000 rpm along with feed rate of 318 mm/min were observed as favorable. Ahmed [2] in 2015, successfully welded 0.44 mm 6xxx Al sheets with butt and lap joint configurations. They calculated the tensile strength in both transverse and longitudinal direction. In 2017, Ahmed [19] published an article on developing a fixture for ultra-thin FSW. Huang [20] in 2018, developed tri-flat pin of tool for 0.5 mm 6061-T4 Al alloy. Zhang [21] in 2019, successfully welded 0.8 mm 1060-H24 Al alloy using a shoulder less tool. Wang [22] in 2019, joined 0.254 mm thick workpiece with micro FSW. Sagher-Abbasi [23] in 2019, formulated an investigation of parametric optimization in case of micro FSW.

Despite of aforementioned studies in the area of micro FSW, there is a need to develop micro FSW for 0.5 mm thick commercial aluminum considering limited investigations on commercial Al with such low thickness. Commercial Al alloy is more challenging as it consists of highest thermal conductivity among all other Al alloys. This in turn leads problems of distortion during weld especially when the thickness is such small of 0.5 mm. It is worth to perform investigation on ultra-thin FSW on commercial Al alloy. In the present investigation, micro FSW was developed to join 0.5 mm commercial Al alloy considering velocity ratios of rotational speed to travel speed.

2. Materials and Method

The micro FSW of commercial Al with thickness of 0.5 mm was performed. Different velocity ratios of 8.91, 12.50, 13.70, 15.30 and 17.30 were varied using combinations of rotational speed and travel speed as shown in Table 1. All the experiments were performed with tilt angle of 1 degree. Baskaro in 2013 suggested that optimal the tilt angle is 1° [15]. The workpiece sheets were clamped on the fixture using copper backing plate, in order to conduct heat using higher conductivity of Cu compare to Al. Ceramic and Copper backing plates showed best result thus copper backing plate was taken into consideration to hinder in the process of heat dissipation [4]. The experimental set-up and tool design (of HSS M-2) are shown in Figure 1 (a) and Figure 1 (b) respectively.

<table>
<thead>
<tr>
<th>Trials</th>
<th>Rotational speed (RPM)</th>
<th>Travel Feed (mm/min)</th>
<th>Speed Ratio</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>1070</td>
<td>120</td>
<td>8.91</td>
</tr>
<tr>
<td>2</td>
<td>1500</td>
<td>120</td>
<td>12.50</td>
</tr>
<tr>
<td>3</td>
<td>1070</td>
<td>78</td>
<td>13.70</td>
</tr>
<tr>
<td>4</td>
<td>765</td>
<td>50</td>
<td>15.30</td>
</tr>
<tr>
<td>5</td>
<td>545</td>
<td>315</td>
<td>17.30</td>
</tr>
</tbody>
</table>

Tensile test coupon was generated for sample by cutting it into required dimension with the help of EDM, according to the ASTM standard as shown in figure1. Fracture surface analysis of the tensile specimen from the weld was carried out using scanning electron microscopy (SEM). Macrostructure and microstructure examination was performed on the welded sample. For that weld specimen was cut transverse to welding with the help of EDM machine, hot mounted followed by grinding and mirror polishing. Macrostructure examination was carried out according to the ASTM-E-381-01, E-340-15 with the utilization of digital camera/stereo microscope. Micro hardness test was performed on the generated weld specimen according to ASTM E-384-17 standards at load of 50gms. The base plate chemical composition along with the mechanical properties were highlighted in table 2 and table 3 respectively.
After welding, the obtained welds were initially inspected with visual examination. Macroscopic analysis, microstructural pattern and mechanical testing were carried out after visual examination for the successfully welded sample. The tensile test coupon was generated by cutting it into required dimensions with the help of wire cut-electro discharge machining (EDM), according to the ASTM standard as shown in Figure 2. Fracture surface analysis for the fractured specimens of tensile testing was carried out using scanning electron microscopy (SEM). Macrostructure and microstructure examination was performed on the cross sectioned sample for the successfully obtained weld. Wire cut EDM machine was used for the same, which was subsequently hot mounted followed by grinding and mirror polishing. Macrostructure examination was carried out according to the ASTM-E-381-01, E-340-15 with the utilization of digital camera/stereo microscope. Micro hardness test was performed on the generated cross sectional weld specimen according to ASTM E-384-17 standards at load of 50 grams (from the middle of the cross section).

3. Result and Discussion

3.1 Joint formation/ Visual Inspection

Table 2 depicts the weld formation scenario with respect to different velocity ratios. Better surface appearance is obtained with velocity ratio of 13.71 having feed rate of 78 mm/min and tool rotation speed of 1070 rpm, whereas difficulties in weld formation were encountered for rest of the combinations. The successful obtained weld with velocity ratio of 13.71 was consists of small defects caused by improper material mixing at the beginning for around 15 mm. The reason of this is non uniform heat distribution across the length of the weld. In the beginning of the weld the heat is subjected to the start position and later same is conducted by the bulk of material across the length and width. Later after 15 mm in the length, excellent material mixing resulted in defect free surface. Besides, the workpiece lift effect was observed in other combinations velocity ratios. This effect was caused due to distortion experienced
during welding phase with aforementioned velocity ratios. 0.5 mm thickness was responsible for high distortion when subjected with inadequate heat input caused by velocity ratio.

Table 2. Welded profile of different parameters and velocity ratio (VR)

<table>
<thead>
<tr>
<th>Experiment No.</th>
<th>Feed Rate (mm/min)</th>
<th>RPM</th>
<th>VR</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>120</td>
<td>1500</td>
<td>12.5</td>
</tr>
<tr>
<td>2</td>
<td>78</td>
<td>1070</td>
<td>13.71</td>
</tr>
<tr>
<td>7</td>
<td>50</td>
<td>765</td>
<td>15.3</td>
</tr>
<tr>
<td>8</td>
<td>31.5</td>
<td>545</td>
<td>17.03</td>
</tr>
<tr>
<td>9</td>
<td>120</td>
<td>1070</td>
<td>8.91</td>
</tr>
<tr>
<td>10</td>
<td>78</td>
<td>765</td>
<td>9.8</td>
</tr>
</tbody>
</table>

3.2 Macrostructure examination

Figure 3 shows macrostructure with different zones for the cross sectional view of experiment number 2 welded with velocity ratio of 13.71. Visual inspection gave a comprehensive quality assurance of the weld obtained, which was found without any defect in the cross sectional view. Thermo mechanically affected zone (TMAZ), Heat affected zone (HAZ), Nugget Zone (NZ)/Weld zone along with the base material were observed in this macro graphic view. The width of NZ was relatively large as compared to other zones due to the conventional design of the tool, which consists of rotation pin. This also ensures adequate stirring action carried out with a very small pin length 0.35 mm in 0.5 mm thick Al alloy.
3.3 Microstructure examination

It can be cross verified from Figure 4 that the discussion presented in previous section with macrostructure of TMAZ, HAZ and NZ was in line with these consisted images. Intense stirring process in NZ resulted in equi-axed fine grain microstructure in stir zone as can be seen from Figure 4 (a)-(c) with different magnification. Dynamic recrystallization of the grain structure was experienced by this NZ. There were no defects presented even at higher magnification in microstructure of the stir zone. HAZ-base material and HAZ-TMAZ can be observed in Figure 4 (d) and (e) respectively. There was a clear distinction between stir zone and HAZ and TMAZ that can be seen comparing internal figures. There were no stirring action presented in the HAZ and TMAZ regions and hence Similar results were obtained in the literature of [20]. TMAZ was formed due to heat and deformation experiences, whereas HAZ was formed due to heat experiences. Relatively larger grains were observed in HAZ compare to TMAZ and NZ because of large experiences of heat dissipation without any stirring effects.
3.4 Mechanical properties

3.4.1 Tensile property

The transverse tension test, where the applied load was orthogonal to the weld direction investigates the joint’s efficiency. Figure 5 exhibits the tensile properties of weld produced with velocity ratio of 13.71. The tensile strength of 115 MPa was obtained, which was noted as more than that of the base metal, wherein the elongation of 34.8% was obtained. Therefore, it can be said that the joint efficiency (i.e. ratio of tensile strength of weld to base material) was 107% of the base metal while the maximum elongation was 80% of the base metal. 80 % elongation shows excellent ductility of welds that increases potential applications in the area of bending and forming industries. In addition, an excellent yield strength of 43.5 MPa was 96% of the base metal obtained. Defect free welding noticed in figure 3 and figure 4 at macro and micro levels respectively supports these excellent strengths. The equi-axed fine grains noticed in the stir zone was responsible for an outstanding ductility. Similar trend was seen in the literature [19], [24], and [20].

Figure 6 shows the comparison between the joint efficiency of the current literature to the other literature as mentioned in the graph. The highest joint efficiency is obtained in the present work that was around 107% whereas the literature of [4] and [20] have 106% and 92% joint efficiency respectively, which shows similar trend. Further, it was observed that in all literatures, represented the yield strength of the weld was decreased than the parent metal. In the present work, the yield strength of 43.5 MPa that was 96% of the base metal was obtained as highest relative to published literature of micro FSW. However, in this work commercial Al alloy was used.
3.4.2 Morphology of tensile fracture surface

The tensile fracture surface morphology was investigated by SEM analysis using Pemtron Make SS-100. Figure 7 shows the fracture surface morphologies analysis of tensile fractured sample under the condition of 13.71-velocity ratio at different magnifications. It can be seen that the fracture was occurred in ductile condition with large amount of dimples on a fractured surfaces. This is in a support of obtained higher elongation from tensile testing.
3.6 Micro-hardness

Hardness tests were performed to check the material’s ability to resist plastic deformation by penetration. As can be seen from Figure 8 that the trend of increasing hardness was observed around micro FSW processed regions. Maximum hardness of 94 HV was noticed at NZ, which was significantly higher than the parent metal. Also, the hardness of HAZ was reported as 70 HV, which was also greater than parent metal. The reason for higher hardness in NZ was fine grain structures. No significant grain coarsening was observed in HAZ that may be the reason in higher hardness of HAZ.
4. Conclusion

In the present work, 0.5 mm thin sheets of commercially pure Al were welded successfully with the help of micro friction stir welding. Following conclusions can be inferred.

- The sound joint of micro FSW on commercially pure Al was achieved at the velocity ratio of 13.71.
- The tensile strength of 100.877 N/mm² and elongation of 23.12% were obtained, which were 90.77% and 51.377% of the parent material respectively.
- Fracture surfaces after tensile testing was observed with elongated dimples indicating ductile fracture.
- The maximum micro hardness of 94 HV was observed in the weld zone.

5. Acknowledgement

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6. References


