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1 Biphenylene Network: A Nonbenzenoid Carbon Allotrope

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11 Abstract:

The quest for planar sp²-hybridized carbon allotropes other than graphene, such as graphenylene and 12 13 biphenylene network, has stimulated substantial research efforts because of the materials' predicted 14 mechanical, electronic, and transport properties. However, their syntheses remain challenging given the 15 lack of reliable protocols for generating non-hexagonal rings during the in-plane tiling of carbon atoms. 16 Here, we report the bottom-up growth of ultraflat biphenylene network with periodically arranged four-, six-, and eight-membered rings of sp²-hybridized carbon atoms, through on-surface inter-polymer 17 18 dehydrofluorination (HF-zipping) reaction. The characterization of this biphenylene network by scanning 19 probe methods reveals that it is metallic rather than a dielectric. We expect the interpolymer HF-zipping 20 method to complement the toolbox for the synthesis of other nonbenzenoid carbon allotropes.

21 One Sentence Summary:

A planar carbon network with four-, six- and eight-membered rings is obtained via an on-surface synthesis
 approach.

24 Main Text:

25 The synthesis of fullerenes (1), carbon nanotubes (2), graphene (3), and cyclo[18]carbon (4) from 26 customized molecular precursors has initiated the era of designer carbon allotropes built from sp, sp², and sp³-hybridized carbon atoms (5). Although numerous sp²-hybridized planar carbon allotropes such as 27 28 graphene (6), pentaheptite (7), and biphenylene network (8) were proposed (Fig. 1A), only graphene has 29 been realized as a purely hexagonal net (3). Progress in the synthesis of other planar 3-coordinate carbon 30 nets (6) with non-hexagons such as 4-, 5-, 7-, 8-membered rings has been very limited. Recently, expanding on the highly successful bottom-up synthesis of benzenoid graphene-related nanostructures 31 32 (9-13), several nonbenzenoid nanographenes (14-17) and nanoribbons (18-20) with embedded 33 nonhexagonal rings were reported.

34 Nevertheless, these carbon nanostructures only exhibit repeating nonhexagonal motifs along one 35 dimension, which has hampered the study of the properties of planar nonbenzenoid carbon allotropes. 36 For instance, the successful synthesis and characterization of a biphenylene polymer chains cannot serve 37 to clarify the theoretical debate on whether biphenylene network is dielectric (21) or metallic (22, 23). 38 Attempts to add nonhexagonal carbon rings along the second dimension through hierarchical 39 polymerization of nonbenzenoid molecules were of limited success. For example, the lateral fusion of 40 polyazulene chains yielded only small segments of phagraphene (24), a planar carbon net with 5-, 6-, and 41 7-membered rings, because the orientation of the azulene moieties could not be controlled (25). Similarly, 42 efforts toward the synthesis of biphenylene network (BPN, Fig. 1B) through sequential deiodinative and 43 debrominative coupling of 1,4,5,8-tetraiodo-2,3,6,7-tetrabromobiphenylene (TTBP, Fig. 1B) failed at the 44 initial polymerization step (26).

45 Here, we demonstrate an alternative approach, in which the nonbenzenoid structural elements were not 46 contained in the precursor, but formed during the lateral dehydrofluorination (HF-zipping) fusion of 47 benzenoid polyphenylene chains. As shown in Fig. 1C, 4,4"-dibromo-2,2',2",5,5',5"-hexafluoro-1,1':4',1"-48 terphenyl (DHTP) first polymerized through debrominative coupling (27) on a Au(111) surface and formed 49 well-aligned assemblies of poly(2,5-difluoro-para-phenylene) (PFPP) chains. In a second step, these chains 50 underwent C-C coupling by HF-zipping (28-31) to form a biphenylene network. The interchain HF-zipping 51 allowed C-C bond formation exclusively between a C-F and a C-H moiety (as marked by cyan ovals in Fig. 52 1C), but not between two C-H or two C-F moieties. As a result, 4- and 8-membered rings formed selectively 53 between the chains. Adequately positioned C-F moieties were essential for the formation of the 54 nonbenzenoid structure. In their absence, such as in the case of plain poly(para-phenylene) chains, 55 dehydrogenative C-C coupling led to the formation of 6-membered rings and thus to regular graphene (32, 56 33).



Fig. 1. Synthetic strategies towards the planar nonbenzenoid biphenylene network (BPN). (A) Lattice structures of three planar carbon allotropes: graphene, carbon pentaheptite (5-7), and biphenylene network (4-6-8). The latter two formally result from graphene by reorganization of bonds (magenta dotted lines). (B) Previous attempts towards the synthesis of BPN from the TTBP monomer. (C) DHTP monomer forming BPN by a two-step sequence consisting of linear polymerization followed by inter-chain HF-zipping.

The monomer DHTP was synthesized in solution through a two-step sequence, as described in supplementary materials (materials and methods), and then vapor-deposited in ultrahigh vacuum onto a clean Au(111) surface held at 300 K. Related scanning tunneling microscopy (STM) images are shown in Fig. S1. Subsequent annealing to 500 K induced the formation of PFPP chains by debrominative C-C coupling. An ordered island of well-aligned PFPP chains (Fig. 2A) exhibited a pronounced zigzag appearance in the zoom-in STM image in Fig. 2D. The periodicity of 8.8 ± 0.2 Å along the chain arose from
alternating up-down tilting of the 2,5-difluoro-para-phenylene (FPP) units, as illustrated by the overlaid
chemical structure on Fig. 2D. This model was confirmed by the non-contact atomic force microscopy (ncAFM) image in Fig. S2B.

72 The aligned PFPP chains fused laterally through HF-zipping to form a biphenylene network during 73 annealing to 680 K, as shown by the STM image in Fig. 2B. This condensation reaction resulted in 74 considerable lateral contraction. The high-resolution nc-AFM images (Fig. 2, E and F) of the regions marked 75 by the white rectangles in Fig. 2B revealed armchair-edged biphenylene ribbons (BPR) with a width of 12 76 and 18 carbon atoms (12-BPR and 18-BPR). These BPRs have alternating H/F edge-termination, rather 77 than pure H termination, as illustrated by the chemical structure in Fig. 2E and confirmed by the simulated 78 AFM images in Figs. S3,4. The widest ribbon (21-BPR) we observed was achieved by further annealing the 79 sample in Fig. 2B to 700 K (Fig. 2C).

80 The selectivity for biphenylene network formation is related to the 2D chirality of the PFPP chains. Only 81 HF-zipping between chains of identical handedness led to 4- and 8-membered rings (4-8 type fusion, 82 labeled in Fig. 2D). Otherwise, 6-membered rings formed (6-6 type fusion, see Fig. 2D), resulting in the 83 formation of an armchair-edged graphene nanoribbon (6-AGNR) with the normal benzenoid topology (Fig. 84 S6). The 4-8 type fusion was overall favored ((2.4 ± 0.3) :1 over the 6-6 type, see supplementary text and 85 Fig. S8 for the yield analysis), because each PFPP domain consisted mainly of chains with identical 86 handedness, e.g., the R_a handedness in Fig. 2D (see also Fig. S5). Therefore, the aggregation of the PFPP 87 chains in almost homochiral domains was essential for the formation of biphenylene network. Furthermore, sufficient length of the PFPP chains is a prerequisite for the formation of wide BPRs, as is 88 89 discussed in supplementary text.



91 Fig. 2. Formation of biphenylene network by two-step polymerization of DHTP. (A,B) Overview STM images 92 taken after deposition of DHTP onto Au(111) at 300 K followed by post-annealing to 500 K (A) and 680 K (B). 93 The white arrow in panel (B) marks a remaining PFPP chain for comparison. (C) nc-AFM image of a wide 94 biphenylene ribbon (BPR) obtained by further annealing of the sample in panel (B) to 700 K. Note that the BPR 95 is jointed with regular graphene at the right side due to minor parasitic 6-6 type fusion. (D) Zoom-in STM image 96 of the PFPP chains in panel (A) overlaid with chemical structures; 2D enantiomers labeled with R_a and S_a. (E,F) 97 Zoom-in nc-AFM images of the BPRs outlined by the white rectangles in panel (B). Panel (E) is overlaid with the 98 chemical structure. Scale bars: (A and B) 5 nm; (C to F) 0.5 nm. (G) XPS spectra for the indicated core levels and

annealing temperatures. (H) Mass spectrometer signals for the indicated *m/z* values recorded during controlled
heating (1 K/s) of bare and monolayer DHTP-covered Au(111).

101 The reaction progress was monitored by x-ray photoelectron spectroscopy (XPS) (Fig. 2G) and 102 temperature-programmed desorption (TPD) mass spectrometry (Fig. 2H). For as-deposited DHTP on 103 Au(111) at 300 K (Fig. 2**G**, 300 K), the F1s (686.8 eV) and Br $3p_{3/2}$ (183.5 eV) binding energies were typical 104 for C-F and C-Br groups (28, 34) and confirmed that carbon-halogen bonds were intact. The C1s signal 105 could be consistently deconvoluted into four components attributed to C-F (pink), C-Br (yellow), C-H (cyan) 106 and C-C (blue) with 1:3:3:2 intensity ratios, in agreement with the composition of DHTP. XPS also 107 confirmed the formation of PFPP chains at 500 K: Lowering of the Br3p_{3/2} binding energy by 2.1 eV 108 indicates C-Br bond dissociation (34), and the C1s signal could be deconvoluted into three components 109 (1:1:1) representing C-F, C-H, and C-C bonds (Fig. 2G, 500 K), in agreement with the chemical structure of 110 the chains. The slight overall shifts (-0.3 eV) of C1s and F1s signals we attributed to a work-function change 111 (35), whereas their attenuation (35% less) was caused by partial desorption of DHTP, as confirmed by the 112 TPD signal of DHTP in Fig. 2H.

113 The gradual attenuation of the F1s and C-F related C1s signals upon further annealing (Fig. 2G, 650 K and 114 700 K) was indicative of partial removal of fluorine from the C-F groups by HF-zipping. The corresponding 115 TPD trace confirmed HF release above 600 K (Fig. 2H and Fig. S9). Note that the C1s spectra at 650 K and 116 700 K showed a component at 283.4 eV, labelled C-Au, which we attributed to benzyne intermediates 117 bonding to the surface after removal of H/F pairs in PFPP chains (Fig. S11). This component decreased 118 from 650 to 700 K because of the progressive fusion of the PFPP chains. Therefore, we tentatively propose 119 the coexistence of HF-zipping (major reaction pathway) and the benzyne-based mechanism (minor 120 reaction pathway) for the lateral fusion of PFPP chains (see supplementary text and Fig. S10).

121

122 We systematically characterized BPRs of different widths with scanning tunneling spectroscopy (STS) and 123 DFT calculations. Figure 3A shows the normalized differential conductance (dI/dV) spectra of six different 124 BPRs with widths from 6 to 21 carbon atoms. The spectra were taken at the positions marked by the 125 colored dots in the AFM images on the left sides of Figs. 3, B to G. The narrower BPRs (6-BPR to 12-BPR) 126 showed distinct valence band (VB) and conduction band (CB) onsets (see also additional d//dV spectra and 127 maps in Figs. S13 and S14). Detecting the resonances in the dI/dV spectra corresponding to the band 128 onsets in the wider BPRs (15-BPR to 21-BPR) was more difficult because of the narrow gaps and extra 129 features around zero bias potentially arising from the Au(111) surface state.

130 We used dI/dV maps (Figs. 3, B to G, middle and right panels, Exp. column) to identity which features in 131 the dI/dV spectra corresponded to states with similar symmetry compared to the VB and CB of the 132 narrower BPRs (see also additional dI/dV spectra and maps in Fig. S15-17). These features matched the 133 simulated dI/dV maps of the VB and CB onsets (Figs. 3, **B** to **G**, middle and right panels, Theory column). 134 For 21-BPR, we could detect orbital features in the dI/dV maps (Fig. 3G and Fig. S17B) at all bias voltages, 135 indicating the closing of the gap corresponding to metallic electronic structure. In addition, based on 136 comparison with simulation, the zero bias dI/dV map corresponded to a mixture of the VB and CB features (arising from states at different momenta, see Fig. S19 for calculated band structures), again consistent 137 138 with the metallic response.



Fig. 3. Electronic properties of BPRs with different widths. (A) Differential conductance (d//dV) spectra (colored curves) taken at the positions marked with corresponding colored dots in the left parts of panels (B-G). The gray curves are reference d//dV spectra taken at clean Au(111). The shaded regions denote the bandgaps between the peaks related to VB and CB onsets. (B-G) nc-AFM images, experimental and simulated d//dV maps at the VB and CB onsets of the 6-BPR (B), 9-BPR (C), 12-BPR (D), 15-BPR (E), 18-BPR (F), and 21-BPR (G). These nc-AFM images are cut-outs of the nc-AFM images in Figs. 2C,E,F and Figs. S12C-E. The simulated d//dV maps are consistent with fluorine substituents at the edges (see Fig. S18 for a detailed comparison), in

agreement with the AFM images. All scale bars: 0.5 nm. The gray scale for all d//dV maps is displayed in the

right part of the panel (G). All dI/dV spectra and maps were recorded with CO-functionalized tips.

149 The bandgap evolution of the BPRs with different widths is summarized in Fig. 4. The measured gaps are 150 overall somewhat larger than the values predicted by DFT at the HSE06 level, but the trend of decreasing 151 bandgaps with increasing widths of the BPRs (Fig. 4, magenta line) agreed well with calculations (black line). This decrease was much steeper than that for the corresponding AGNR series reported previously 152 153 (Fig. 4, blue line) (33, 36). In contrast to the benzenoid AGNRs, the bandgap of the nonbenzenoid BPRs 154 closed completely between 18-BPR and 21-BPR. According to the DFT calculations, the edge termination 155 has no substantial influence on the bandgaps of the BPRs (Fig. S19). This result was also consistent with 156 the predicted bandgaps of pure H-terminated BPRs in previous work (22) as illustrated by Fig. S20. The 157 rapid decrease of the measured gaps (from 2.35 to 0 eV) of biphenylene networks with increasing sizes 158 provided strong evidence that this material had metallic character, rather than being dielectric with a gap 159 of 2 eV, as was reported in a previous theoretical study (21).



Fig. 4. Comparison between experimental (BPR_Exp., magenta line) and calculated (BPR_Cal., black line)
bandgaps of BPRs with AGNRs (AGNR_Exp., blue line) of the same width, as given by the number of carbon
atoms, N_a. The AGNR values are adapted from ref. [33] on Au(111) and ref. [36] on AuSIL/Au(111).

164 Our interpolymer HF-zipping technique for the formation of rings with nonbenzenoid topology in a 165 periodic arrangement paves the way for the exploration of new planar sp^2 carbon allotropes and their 166 properties. Further synthesis progress will focus on increasing yield and selectivity, where the on-surface 167 approach can offer distinct advantages over solution-based reactions (37). The HF-zipping reaction is also 168 expected to work on metal-oxide surfaces (28, 30), offering decoupling from the metallic substrate. 169 Further direct potential for applications arises from transfer to other substrates or media. For instance, 170 nonbenzenoid 2D carbon allotropes with large rings (> 6) may represent superior anode materials in 171 lithium-ion batteries because of their predicted larger lithium storage capacity compared to graphene (38). 172 The revealed metallicity of biphenylene network also makes it a candidate for conducting wires in future 173 carbon-based circuitry.

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295 Supplementary Materials

- 296 Materials and Methods
- 297 Supplementary Text
- 298 Figs. S1 to S24
- 299 References (39-51)