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Solubility of Chromium in DON Smelting

Katri Avarmaa 🝺^a, Dan Strengell^{a,b}, Hannu Johto^{c,d}, Petri Latostenmaa^c, Hugh O'Brien 📭^e, and Pekka Taskinen 🍺^a

^aDepartment of Chemical and Metallurgical Engineering, Aalto University, School of Chemical Technology, Espoo, Finland; ^bGasum Oy, Espoo, Finland; ^cBoliden Harjavalta, Harjavalta, Finland; ^dMetso Outotec, Espoo, Finland; ^eGeological Survey of Finland, Espoo, Finland

ABSTRACT

Chromia-bearing raw materials in nickel and copper matte smelting are difficult to process due to their tendency of forming solid chromite spinel precipitates, leading to formation of mushy slag and buildups in the smelting vessel. The solubility of chromia in smelting slags, and especially in mattes, are not known accurately and new data for iron-silicate slags in equilibrium with low-iron nickel mattes have been measured at 1350–1450°C. Typical copper-bearing nickel mattes with Ni:Cu \approx 2 (w/w) in the DON (Direct Outotec Nickel) process with 2 to 10 wt% [Fe]_{matte} have been equilibrated in carefully controlled S₂-O₂-SO₂-Ar gas atmosphere experiments with the corresponding silica saturated iron-silicate slags. The phase assays post quenching were measured by electron probe X-ray microanalysis, including the molten slag and matte as well as the solid phase of chromite spinel. Laser ablation ICP mass spectrometry was used to measure the trace elements in the matte. An additional variable in the slag composition was magnesia concentration, varying from zero to 10 wt% (MgO)_{slag}. The solubility of chromium in the slag at 1400°C was \approx 0.7 wt% (Cr) and in the nickel matte 30–100 ppm [Cr], depending on the iron concentration of the sulfide matte. The impact of MgO on the chromium concentration in slag was small and within the experimental error of the measurements.

KEYWORDS

Nickel; sulfide matte; slag; thermodynamics

Introduction

Nickel matte smelting is chemically close to copper matte smelting and their fluxing practices are essentially the same (Font, Hino and Itagaki 2001; Kellogg 1987). Nickel sulfide ores are generally mineralogically and geochemically distinct from copper sulfide ores. Besides the often lower content of sulfides, the nickel concentrates contain silicate gangue materials rich in MgO and in certain cases also Cr_2O_3 , i.e. traces of ferrochromite (Cramer 2001). These features make the primary smelting operation very demanding, due to the higher liquidus temperature of the slag and new dense compounds becoming stable in the operation and deporting as solid precipitates on the matte-slag interface in the smelting vessel (Barnes and Newall 2006; Nell 2004; Ritchie and Eksteen 2011).

Limited experimental studies have been conducted to investigate chromium behavior in iron-silicate slags and matte systems in nickel and copper matte smelting conditions. Kwatara (2006) studied chromium in copper matte – slag system under various pS₂ and pO₂ conditions at 1300–1500°C, whereas Du Preez (2009) equilibrated Cu-Ni-Fe-S matte and iron-silicate slag with different additives (Al₂O₃, MgO, CaO) with pO₂ = 10^{-8} atm and pS₂ = $10^{-3.5}$ atm at 1550 °C. Additionally, De Villiers and Kleyenstüber (1984) investigated chromium equilibrium in iron matte – alumina/magnesia-containing ironsilicate slags at 1450°C. Wright, Jahanshahi and Sun (2009) instead studied the chromium distribution behavior in the matte – Pt-alloy system under reducing conditions of fixed pS₂ = $2.5x10^{-5}$ atm and varying pO₂ = 10^{-15} – 10^{-12} atm at 1400°C. Sukhomlinov et al. (2020) measured chromium dissolution in alumina/lime-containing iron-silicate slags under black copper smelting conditions at 1300°C. Henao, Hino and Itagaki (2001) determined the chromium distribution between a nickel alloy and FeO_x-MgO-SiO₂ slags at 1500–1600 ° C at MgO saturation and calculated the activity coefficient of chromia in the slag.

In PGM extraction, high MgO and Cr₂O₃ concentrations in iron-nickel-copper concentrates of the Bushveld Complex require higher smelting temperatures than nickel matte smelting, close to 1600°C (Andrews 2005; Coetzee 2006; Eksteen, van Beek and Bezuidenhout 2011; Nell 2004; Nelson et al. 2019). This is most likely the case also elsewhere where the ore body contains chromite minerals (Maier and Groves 2011). Despite the industrial importance, very little fundamental data are available in the literature on the chemistry and phase equilibria of chromia in iron-silicate slags, as well as the combined effect of chromia and magnesia in smelting slags. Murck and Campbell (1986) studied the dissolution of chromium in iron-silicate slags at 1300-1500° C under various oxygen partial pressures. Similarly, Roeder and Reynolds (1991) did experimental research for basaltic melts at chromite saturation and as a function of oxygen partial pressure between 1200 and 1400°C. Eguchi, Uchida and Chiba (1977) measured the solubility of chromium in a solid iron - ironsilicate slag system at 1230-1400°C, providing solubility of Cr₂ O₃ as 1.5, 3 and 5 wt% at 1230, 1300 and 1400°C, respectively. In the study by Nell (2004), the chromia solubility in iron-silicate slag increased at low pO_2 , being >5 wt% (Cr₂O₃) with pO_2 < 10⁻¹¹ atm at 1550 °C. In addition, Chen, Jak and Hayes (2005) investigated the liquidus line at 1400-1600°C with 2 wt%

CONTACT Katri Avarmaa 🖾 katri.avarmaa@gmail.com 🗈 Department of Chemical and Metallurgical Engineering, Aalto University, School of Chemical Technology, Aalto 00076, Finland.

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 Cr_2O_3 at silica saturation. Ilyushechkin, Hayes and Jak (2015) determined phase equilibria in silica-spinel and silica-spinelpyroxene saturation under air and reducing conditions at 1350-1490°C. They presented chromia solubilities between 0.2 and 0.5 wt% in the oxide slag at the oxygen partial pressures $(7.8*10^{-7}-2.7*10^{-4})$ generated with the volumetric ratio CO₂:H₂ = 132:1. According to Sommerville et al. (2004), a spinel of (Mg,Fe)(Al,Cr,Fe)₂(O)₄ was precipitated and the solubility of chromium in the molten silicate was 0.3-1.3 wt% Cr at 1350-1630°C. They reported also chromium bearing olivine (Fe,Mg,Cr)₂SiO₄ and pyroxene (Fe,Mg,Cr)SiO₃ to be stable below 1450°C as primary phases. In reducing conditions and iron-free slags related to ferrochromium smelting, intensive experimental and modeling studies have been conducted in the literature, for example Curr, Wedepohl and Eric (1988), Xiao and Holappa (1993), Pan and Eric (2004), Arnout et al. (2009) and Albertsson et al. (2013). Several papers are also available on the Cr²⁺/Cr³⁺ and Cr³⁺/Cr⁶⁺ equilibria in various slags (Farges 2009; Hanson and Jones 1998).

The aim of the present study was to determine the equilibrium solubility of chromium in DON (direct Outotec nickel) slags at typical iron concentrations (2–10 wt%) of the DON matte (Mäkinen and Ahokainen 2009) and to quantify the distribution of chromium between the matte and iron-silicate sags with varying MgO concentrations at silica and chromite spinel saturations. The measurements were carried out at a fixed SO₂ partial pressure of 0.1 atm that is generally used in the matte-slag studies and is based on the sulfur-oxygen potential diagram, i.e., Yazawa diagram (Yazawa 1974). The temperature range between 1350 and 1450°C was selected based on industrial practices of nickel matte smelting where the magnesia concentration of the feed mixture and thus that of the slag determines the practical slag tapping temperature (Mäkinen and Taskinen 2008).

Materials and methods

The experimental techniques included a high-temperature equilibration at controlled temperature and CO-CO₂-SO₂-Ar gas atmosphere, quick drop-quenching to an ice water mixture and direct phase analysis by electron probe X-ray microanalysis (EPMA) and laser ablation high resolution inductively coupled mass spectrometry (LA-HR-ICP-MS). Nickel matte was equilibrated with iron-silicate slag in silica crucibles in a temperature range of 1350–1450°C and matte grade range of 2–10 wt% [Fe]. This paper is a continuation of our previous work (Strengell et al. 2016) where we presented and discussed the results for nickel and matte phases in the DON process. The same experimental technique and assembly were used in this study as those described in detail in Strengell et al. (2016).

The experiments were carried out in a vertical tube furnace (Lenton PTF 15/45/450) employing controller unit of Eurotherm 3216 PID. The temperature was followed and measured during the experiments next to the sample with a calibrated S-type Pt/90%Pt-10%Rh thermocouple (Johnson-Matthey, UK; accuracy \pm 2°C) and the ambient room temperature with a calibrated Pt100 sensor (Platinum Resistance Thermometer; SKS Group, Finland).

An equal amount (0.05 g/each) of premixed Cu₂S-FeS-Ni₃S₂ matte and Fe₂O₃-SiO₂-MgO-Cr₂O₃ slag were equilibrated in silica crucibles under controlled partial pressures of pS₂, pO₂ and pSO₂ at fixed temperature. All the aforementioned materials used were commercially available pure, high-grade reagents (>99.5%) supplied by Umicore, Alfa-Aesar or Sigma-Aldrich. The partial pressure of sulfur dioxide was held constant at 0.1 ± 0.01 atm and the partial pressures of O₂ and S₂ were fixed to a certain target 'matte grade' in each experiment, calculated using MTDATA software (databases of MTOX and SGTE pure substance) (Davies et al. 2002), see Table 1 for the used gas flows and corresponding pO_2 and pS₂. Inert argon was added to the gas mixture in order to increase the flow rate and N₂ post-experiment for flushing. The reactive gases of CO₂, CO and SO₂ were controlled by accurate mass-flow controllers (Aalborg DFC26, USA, accuracy of ±1% from full range) and were also calibrated employing a bubbleo-meter device (Dublin, OH). The inert gases of Ar and N2 were controlled by rotameters (Aalborg 052–01-SA, USA; accuracy \pm 2%). The use of silica crucible created silica saturation and the addition of Cr2O3 was 2 wt% (0.002 g/sample) to ensure spinel saturation. A 4-h equilibration time was shown to be sufficient to reach equilibrium (results presented in (Strengell et al. 2016)) and thus, employed in the experiments. After 4-h equilibrium the sample was quickly quenched and prepared by using metallographic techniques for preliminary SEM-EDS and primary EPMA and LA-HR-ICP-MS analyses.

Two identical experimental series were carried out to determine the concentration of chromium in slag and matte, as well as the matte-slag distribution coefficients. Each series contained nine experiments at 1400°C (three matte grades with three MgO concentrations), and three experiments at 1350 and 1450°C, respectively, at a constant 5 wt% [Fe] in matte with three MgO concentrations. In the following sections, parentheses () and [] refer to the slag and matte phases, respectively, except when examining the spinel result that are also presented with () parenthesis.

The primary phase analyses were executed using a Cameca SX100 EPMA (Cameca SAS, France) with wavelength dispersive spectrometers (WDS) housed at Geological Survey of Finland. The EPMA settings used, detection limits and standards have been presented earlier in detail (Strengell et al. 2016). The detection limit of chromium in the EPMA measurements were 125, 109, and 199 ppmw in matte, slag and spinel, respectively. The laser ablation single collector ICP-MS

 Table 1. The used gas flows and their partial pressures at each experimental temperature.

Temp/ °C	Target Fe in matte/ wt-%	MgO in slag/wt- %	log P _{O2} / atm	log P _{S2} / atm	CO/ mL min ⁻¹	CO ₂ / mL min ⁻¹	SO ₂ / mL min ⁻¹	Ar/ mL min ⁻¹
1350	5.0	0, 5 and 10	-7.25	-3.17	8	55	40	300
1400	2.5	0, 5 and 10	-6.74	-3.41	8	55	40	300
1400	5.0	0, 5 and 10	-6.91	-3.15	9.9	55	40	300
1400	10.0	0, 5 and 10	-7.18	-2.67	16	55	40	300
1450	5.0	0, 5 and 10	-6.67	-3.03	13.3	55	40	300

analyses of the mattes were executed by using a Nu AttoM SC-ICP-MS (Nu Instruments Ltd., Wrexham, UK) and an Analyte G2 193 nm ArF laser-ablation system (Teledyne CETAC Technologies, Omaha, USA). The laser was run at a pulse frequency of 10 Hz and a pulse energy of 4 mJ at 40% attenuation to produce an energy flux of 3 J/cm² on the sample surface. Mass1 and ⁵⁷Fe was used as external and internal standards, respectively, for quantification, and BCR2G and BHVO-2 G for quality control. The measurements were performed for Mg, Si, S, Cr, Ni and Pb at low resolution (M/ Δ M = 300) using the fast scanning mode. The instrumental uncertainty of the LA-HR-ICP-MS technique was evaluated as ± 5–10 wt%. Eight analyses points per phase were measured with both analytical techniques, and the averages and standards deviations of those were calculated and used as results.

Results

The obtained chromium concentration of the slag at 1400°C as a function of iron concentration of the nickel matte is shown in Figure 1. The data were calculated averages of EPMA results from the glassy and originally molten part of the silicate slag.

The scatter and uncertainty of the observations were relatively large and it is difficult to conclude whether the oxygen partial pressure or MgO concentration in slag have an impact on the chromium solubility in the composition range investigated. At least the influence of MgO over the studied concentration rage was small.

The temperature dependence of chromium solubility at silica-spinel saturation is shown in Figure 2. The measured solubility tended to increase slightly with temperature, in spite of the severe increase in silica concentration which lowered the solubilities of several other metals in iron-silicate slags, as indicated by Takeda (1997) and Park et al. (2011). This was particularly true in high-iron slags with less than 30 wt% (SiO₂), i.e. on the basic regions away from the orthosilicate composition of the Fe-O-SiO₂ system. Again, experimental scatter was relatively large.

The saturation phase of chromia in the slag was a spineltype solid solution in all MgO concentrations of the slag. It had higher density than the molten bulk slag which characterized its deportment in smelting vessels (Barnes and Newall 2006). The composition of the spinel as a function of the prevailing oxygen partial pressure and MgO activity of the slag changed as magnesia dissolved into the spinel from the slag. EPMA analyses indicated that the spinel did not contain copper oxide or silica. In the case of the alumina-bearing slags, the equilibrium spinels would dissolve alumina (Ilyushechkin, Hayes and Jak 2015; Klemettinen, Avarmaa and Taskinen 2017) but due to the low alumina levels in the DON slag, the synthetic slags used in this study and the spinels were free of alumina.

Distribution of chromium between spinel and slag is shown in Figure 3 at different magnesia concentrations at 1400°C and at different temperatures with about 5 wt% MgO in the slag. It was defined in the conventional way as:

$$L^{sp/s}Me = (wt\%Me)_{spinel} / (wt\%Me)_{slag}$$
(1)

where Me denotes the dissolved metal in the spinel (sp) and molten silicate slag (s) phases, respectively. It is a thermodynamic variable which is independent of the scale of the system and thus particularly suitable for industrial use. This thermodynamic relation (1) also results in an estimate of the uncertainties of the obtained distribution coefficients as:



Figure 1. The obtained solubilities of chromium in molten iron-silicate slags at silica-spinel saturation as a function of iron concentration of nickel matte at 1400°C; the error bars show the standard deviations of the results with MgO = 0 in the slag and the red horizontal line shows the average value of all experimental points of 0.66 wt % (Cr).



Figure 2. The impact of temperature on the chromium solubility in silica-spinel saturated, molten iron-silicate slags at 1350–1450°C and 5 wt% [Fe] in the DON matte; the trend line has been fitted to the points with 5 wt% (MgO).



Figure 3. Distribution coefficient of chromium between spinel and molten slag (a) at 1400°C as a function of the iron concentration of the DON matte and (b) at 5 wt% MgO in the slag at 1350–1450°C; the error bars in (a) refer to the experimental data at 0 wt% MgO in the slag and in (b) at 1350°C. The red horizontal lines show the average of the experimental points presented in each diagram.

$$\Delta L = L \cdot \left\{ \Delta (Me)_{sp} / (Me)_{sp} + \Delta (Me)_{slag} / (Me)_{slag} \right\}$$
(2)

where $\Delta(Me)_{sp}$ is the standard deviation of EPMA measurement of the spinel and $\Delta(Me)_s$ is that in the slag, and L, $(Me)_{sp}$ and $(Me)_s$ are the distribution coefficient, solubility of Me in spinel and in slag, respectively.

The distribution coefficient obtained indicated that chromium was preferentially deporting in the solid spinel and thus it contained much more chromium than the average slag. No significant dependence could be seen between the distribution coefficient and iron concentration of the matte, nor the MgO concentration in slag and temperature.

The LA-HR-ICP-MS results of the matte domains allow an estimation of the chromium solubility in the nickel matte at silica-spinel saturation, Figure 4. The dissolved chromium concentrations in the matte were below the detection limit of EPMA but were at suitable levels for measurement by LA-HR-ICP-MS techniques as presented in Figure 4.

The only available data on chromium solubility in matte concerning slag-matte equilibrium in the literature are the ICP and EDS measurements by Kwatara (2006). Their values



Figure 4. Measured solubilities of chromium in nickel mattes at 1350–1450°C analyzed by LA-HR-ICP-MS as a function of iron concentration of the DON matte analyzed by EPMA; symbols refer to various MgO concentrations of the slag (… MgO = 0, -MgO = 5,10 wt%).

ranged from 0.3 wt% [Cr] to about 0.6 wt% [Cr] at 1300–1500° C by ICP analyses and with EDS they measured concentrations from 0 to 1 wt% [Cr] depending on the analyzed 'sub-phase' of the matte and the experimental conditions. The ICP-based concentrations were significantly higher than measured in the present study by the LA-HR-ICP-MS technique, whereas part of the EDS results by Kwatara (2006) were below the detection limits and could be in the same range as our results. The ICP technique requires manual phase separation and thus increased risk for distorted results as described by Fallah-Mehrjardi et al. (2017). The effect of MgO concentration in the slag seemed to

have a small impact on the chromium solubility in the matte. Nevertheless, MgO had much greater influence on the activity of iron and thus on the composition of matte in the system.

The obtained matte-to-slag distribution coefficient of chromium is shown in Figure 5 at various MgO concentrations. The error bars were calculated in a similar way as presented in the distribution data for chromium between spinel and slag phase, according to Equation (2). As a function of iron concentration of matte, the distribution coefficient of chromium L^{m/s}Cr increased slightly which indicated a small tendency for chromium to increase its solubility in the matte when oxygen pressure was lowered and sulfur pressure increased.

The current results indicated that from industrial point of view, the solubility of chromium in the sulfide matte of nickel smelting is negligible, and practically all chromium in the feed mixture of the primary smelting vessel in the DON process is deported in the slag, either as dissolved chromium oxide in the silicate slag or as high-density solid chromite. It is obvious that the composition of chromite spinel is a function of the spinel forming from the present slag components, such as alumina, nickel oxide, zinc oxide or magnesia (De Wilde et al. 2015).

The lattice structure of the spinel solution phase can be expressed as $(Me)(M)_2(O)_4$ where Me are predominantly divalent cations and M predominantly trivalent. If the deviations from cation:anion ratio of 3:4 are taken into account, the sublattice model of the spinel solution includes interstitial cations and vacancies in one or several cation sublattices (Barry et al. 1992). In this case, a simplified version was applied due to the relatively low experimental temperature where no major deviations from cation stoichiometry of 3:4 exist. The distribution of cations in spinel was calculated from mass balances, assuming that Mg and Ni are predominantly in the normal spinel location, i.e. in the first sublattice, and chromium in the second (Farges 2009). The results at 1400°C are shown in Figure 6 as a function of magnesia concentration in the slag. One can see that the dissolution of magnesia from the



Figure 5. The distribution coefficient of chromium between nickel matte and iron-silicate slag at 1350–1450°C as a function of iron concentration of the DON matte; the various MgO concentrations of the slag cannot be distinguished from each other due to scattering in the distribution coefficient values.



Figure 6. The estimated distributions of cations in the spinel (Me)(M)₂(O)₄ sublattices as a function of magnesia concentration of the bulk slag at silica-spinel saturation at 1400°C.

silicate into the spinel lattice gradually pushed iron from the inverse spinel location (Fe,M) in the octahedral sites to tetrahedral sites (Fe,Me).

Discussion and conclusions

The experimental results obtained in this study indicated that chromium (oxide) solubilities are small in iron-silicate slags at matte smelting temperatures and conditions used in nickel and copper smelting. An average concentration value of 0.66 wt% (Cr) or 0.96 wt% (Cr₂O₃) dissolving in the slag at double saturation of chromite spinel and silica at 1400°C was obtained in the current study. Although some experimental points included relatively big scatter, the overall trends and distribution coefficient results were very consistent providing reliability for the results.

The matte-slag equilibrium results obtained in this study were much lower than presented by Kwatara (2006) at 1300–1500°C. The distribution coefficients $L^{m/s}$ based on ICP measurements in Kwatara's study were between 0.07 and 0.13 depending on temperature, pO₂ and pS₂.

Bartie (2004) measured chromium distributions in the slag phase between different equilibrium phases at 1400–1600°C as a function of oxygen pressure. His slags contained about 5 wt% alumina and lime, and their magnesia concentrations varied from 15 to 20 wt% MgO. His results for chromium solubilities in slags and the distribution coefficients between spinel and slag are in good agreement with the present data at 1400°C. Roeder and Reynolds (1991) measured the solubility of chromium in a molten silicate at chromite saturation and as a function of oxygen partial pressure at 1200–1400°C. Their silicate was a low-iron, sulfur-free basalt, high in alumina and silica. The obtained chromium solubilities clearly increased with decreasing oxygen pressures and were clearly lower, about 0.1-0.2 wt% (Cr₂O₃) at pO₂ = 10^{-7} - 10^{-8} atm and 1400° C. Sukhomlinov et al. (2020) reported chromium solubilities of 0.1–0.25 wt% at 1300°C and oxygen partial pressures of pO₂ = 10^{-5} - 10^{-9} atm in ternary Fe-O-SiO₂ slags at silica-spinel saturation that are slightly on the lower side compared with the results achieved in this study.

The small distribution coefficient of chromium between the nickel matte and molten silicate slag at 1350-1450°C suggests that the expected carryovers of chromium from an electric furnace (EF) to the converting stage (e.g. Eksteen, van Beek and Bezuidenhout 2011) are due to slag and not matte transferred from EF to converters. The entrainment of slag into matte during tapping should then be avoided more carefully than today's practices permit. The current results also make possible to estimate the maximum chromium permitted in the feed mixture in order to avoid the formation of the buildups on the furnace walls and heart (Lennartsson et al. 2015). Due to the operational windows of copper matte smelting being close to that of DON smelting (Mäkinen and Taskinen 1998; Matousek 2014), the results obtained in the present study can be applied also in copper smelting operations.

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Disclosure statement

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

ORCID

Katri Avarmaa (http://orcid.org/0000-0002-8632-3788 Hugh O'Brien (http://orcid.org/0000-0002-7900-579X Pekka Taskinen (http://orcid.org/0000-0002-4054-952X

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